钛合金电子束焊熔宽 PID 控制系统建模与仿真

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摘 要:由于焊接过程的复杂性,因此很难得到熔池动态过程的解析数学模型,使得焊接过程建模成为焊接过程控制的一大难题。为了建立电子束焊熔池动态过程模型,采用阶跃响应法确定出了钛合金电子束焊电子束流与熔宽的数学模型的类型和结构,计算出了该模型的参数,建立了电子束流与熔宽的传递函数模型。利用获得的电子束流与熔宽的数学模型,采用 Z-N 经验公式法确定出了 PID 控制器参数,在此基础上利用MATLAB/SIMULINK 建模与仿真平台建立了钛合金电子束焊熔宽 PID 控制系统模型,并对该模型进行了仿真试验。结果表明,所设计的控制系统动态性能和稳态性能良好。

关键词: 电子束焊;熔宽;建模;仿真

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0 序 言

实现焊接自动化是焊接技术的发展趋势之一^[1] ,真正意义上的自动化焊接 ,其核心问题仍然是实现对焊接熔池动态过程的自动控制^[2] . 经典控制系统首先要求建立被控对象的数学模型 ,在此基础上才能进行 PID 控制系统的设计^[3] ,即便是近年来发展起来的智能控制理论. 尽管在一定程度上不依赖于精确的数学模型 ,但有尽量多的被控对象动态特性的知识对建立任何一种智能控制系统来说都是有帮助的. 但是由于焊接过程是多参数强耦合非线性时变的过程 ,因此很难得到熔池动态特征的解析模型 ,使得焊接熔池动态过程的实时控制以及由此决定的焊接质量控制成为困扰焊接界的一大难题^[4,5] ,也是实现焊接自动化的最大障碍. 为了实现焊接过程自动控制 ,对焊接动态过程有一个系统的认识是十分必要的.

文中采用阶跃响应法确定出钛合金电子束焊电子束流与熔宽的数学模型的类型和结构,并且计算出了该模型的参数,在获得被控对象数学模型的基础上,利用 MATLAB/SIMULINK 建模与仿真平台建立了钛合金电子束焊熔宽 PID 控制系统模型,并对该模型进行了仿真试验,试验结果表明所设计的控

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制系统动态性能和稳态性能良好.

1 电子束流与熔宽数学模型的辨识

1.1 电子束流对熔宽的影响规律

为了找到电子束流对熔宽变化的影响规律,在 100 mm×50 mm×20 mm 的 TA15 钛合金平板上进行了堆焊试验,在其它参数一定的情况下,改变束流大小进行焊接,定义焦点在工件表面以上时为正,在工件表面上时为零,在工件表面以下时为负.具体试验条件如表1所示.

表 1 堆焊试验条件
Table 1 Experimental conditions of surfacing

编号	工作距离 d/mm	加速电压 U/kV	电子束流 I/mA	焊接速度 v/(mm•s ⁻¹)	焦点位置 f/mm
1	175	55	20	6	0
2	175	55	30	6	0
3	175	55	40	6	0
4	175	55	50	6	0

熔宽随电子束流的变化如图 1 所示. 由图 1 可知熔宽随着电子束流的增大而增大. 这主要是由于电子束流增大导致工件热输入增加,因而熔宽也相应增大.

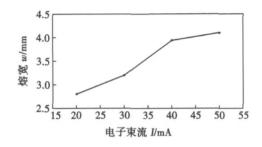


图 1 电子束流对熔宽的影响

Fig. 1 Influence of electron beam to molten pool

1.2 阶跃响应试验

以上的研究仅仅是在试验的基础上对电子束流影响熔宽的规律进行了定性的分析,为了定量分析熔宽在给定的电子束流作用下的变化规律,建立熔宽动态过程数学模型,文中设计了阶跃响应试验. 具体试验条件如表2所示.

表 2 阶跃响应试验条件

Table 2 Experimental conditions of step response

 焊接速度	焦点位置	加速电压	工作距离
$v/(\text{mm} \cdot \text{s}^{-1})$	f/mm	U/kV	d/mm
6	0	55	175

在焊接过程中保持其它条件不变 ,焊接电流以 20 mA 稳定运行 6 s 后突变为 40 mA 经过一段时间 后 焊接过程进入新的稳态 ,试验进行到 15 s 时结束. 同时在焊接过程中利用熔池图像视觉传感系统实时采集熔池图像 ,经图像处理后得到熔宽的变化数据. 图 2 是焊接过程中熔宽的动态变化曲线 ,由图 2 可知 在焊接过程中熔宽的动态变化曲线 ,由图 2 可知 在焊接过程中熔宽始终存在微小波动 ,因此定义稳态输出值的±2% 为一允许误差范围 ,当熔宽达到并保持在这一允许误差范围内 ,就认为它已经达到了稳态输出值. 其阶跃响应曲线如图 3 所示. 需要指出的是 ,图 3 所示的阶跃响应曲线是以初始稳态工作点为坐标原点的增量变化曲线.

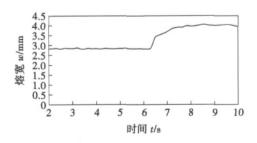


图 2 熔宽动态变化曲线

Fig. 2 Dynamic curve of molten width

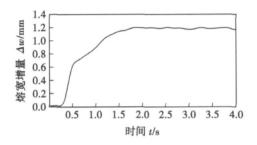


图 3 阶跃响应曲线 Fig. 3 Curve of step response

1.3 模型类型和结构辨识

$$G(s) = \frac{K}{T_{s+1}} e^{-\tau s} \tag{1}$$

式中: K 为放大系数; T 为时间常数; τ 为滞后时间; s 为拉氏算子.

当模型类型和结构确定后,下一步工作就是采用合适的方法对模型进行参数估计,即求得自衡非振荡模型中的参数 K, T 和 τ .

1.4 参数辨识

放大系数 K 可以采用如下算法求得. 对于式 (1) 所描述的一阶惯性加纯滞后环节 ,当阶跃输入幅值为 $x_a = 20$ 时 ,阶跃响应为

$$y(t) = \begin{cases} Kx_o \left(1 - e^{-(t-\tau)/T}\right), & t > \tau \\ 0, & t \leq \tau \end{cases}$$

$$= \begin{cases} 20K \left(1 - e^{-(t-\tau)/T}\right), & t > \tau \\ 0, & t \leq \tau \end{cases}$$

$$0, & t \leq \tau$$

$$(2)$$

需要指出的是 式(2) 是在初始稳态输出值基础上的增量表达式 t 为时间.

对于式(2) 当 $t \rightarrow \infty$ 时,有

$$y(t) \mid_{t \to \infty} = y(\infty) = Kx_0 \tag{3}$$

式(3) 中 y(∞) 是最终稳态输出值 ,由式(3) 可得

$$K = \frac{y(\infty)}{x_o} = \frac{1.198}{20} = 0.0599$$
 (4)

求取 T 和 τ 的过程如下. 为了计算方便 ,首先将式 (2) 转化成无量纲形式 ,为

$$y_o(t) = \frac{y(t)}{Kx_o} = \frac{y(t)}{y(\infty)}$$

$$= \begin{cases} 1 - e^{-(t-\tau)/T}, & t > \tau \\ 0, & t \leq \tau \end{cases}$$
(5)

相应的阶跃响应曲线也转化成无量纲形式,如图 4 所示. 当过程输出在稳态输出允许误差范围内时,表明输入的变化对输出无影响,输出仍然是稳态值,而当过程输出在稳态输出允许误差范围外时,则表明输入的变化对输出产生影响,输出不再是稳态值,而是对输入的变化有了响应. 因此,滞后时间 τ 可以根据阶跃响应曲线超出允许误差范围的时刻(即开始出现变化的时刻)来确定,即

$$\tau = 0.2406$$
 (6)

令 t>τ ,由式(5) 可得

$$y_o(t) = 1 - e^{-(t-\tau)/T}$$
 (7)

取 t=1.08 则 $y_o(t)=0.8$,由式(7) 可得

$$T = \frac{(\tau - t)}{\ln[1 - y_o(t)]} = 0.521 6$$
 (8)

将以上参数值代入式(1) 可得钛合金电子束焊电子 束流与熔宽的传递函数为

$$G(s) = \frac{K}{T_{s+1}} e^{-\tau s} = \frac{0.0599}{0.5216s+1} e^{-0.2406s}$$
 (9)

辨识所得模型计算得到的阶跃响应曲线和实测数据的对比如图 5 所示. 从图 5 中可以看出,辨识所得传递函数的阶跃响应过程和实际过程基本接近,两者之间的误差较小,证明建立的数学模型是准确的.

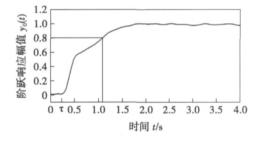


图 4 阶跃响应曲线的无量纲形式

Fig. 4 Dimensionless form of step-response curve

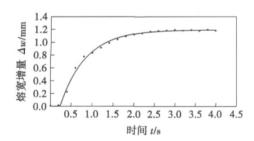


图 5 辨识所得模型计算得到的阶跃响应曲线和实测数据的对比

Fig. 5 Contrast of calculated step-response curve from model obtained by identification and measured data

2 控制系统建模与仿真

2.1 控制系统模型的建立

利用获得的电子束流与熔宽的数学模型,通过 MATLAB/SIMULINK 建模与仿真平台建立钛合金电 子束焊熔宽 PID 控制系统模型.

PID 控制器是一种线性控制器 ,它将设定值与系统输出值的偏差作为输入变量 ,通过调节比例 (P)、积分(I)、微分(D) 三个环节对被控对象进行控制. 控制器的输入变量为熔宽的偏差 ,输出变量为电子束流的变化量. 在 PID 控制器中 ,比例系数 K_P 、积分时间常数 T_I 和微分时间常数 T_D 三个参数是决定控制精度的关键 ,目前 PID 控制器参数整定方法很多 ,文中选择经典的 Z-N 经验公式法计算 K_P , T_I 和 T_D 的值. 具体计算过程如下:

$$K_{\rm p} = \frac{1.2T}{K\tau} = 43.465$$
 (10)

$$T_i = 2\tau = 0.4812$$
 (11)

$$T_{\rm d} = 0.5\tau = 0.1203$$
 (12)

在 PID 参数确定后 就可以建立如图 6 所示的钛合 金电子束焊熔宽 PID 控制系统模型.

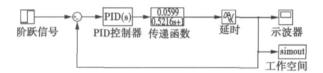


图 6 钛合金电子束焊熔宽 PID 控制系统模型

Fig. 6 PID control system model of molten width of titanium-alloy electron-beam welding

2.2 仿真试验

为了检验所建控制系统模型的动态性能和稳态性能,用建立的模型进行了仿真试验. 具体作法是将一个单位阶跃信号作用于控制系统输入端,观察系统的输出随时间而变化的情况,根据系统响应对控制系统模型的动态性能和稳态性能作出判断. 之所以选择阶跃信号作为输入,一方面是因为它容易产生,系统求解简单;另一方面是因为当它作用于给定值输入端时可以代表给定值突变的情况,此时的系统响应若满足要求,则在实际干扰作用下一般也会满足要求.

图 7 为控制系统模型的单位阶跃响应曲线 ,其上升时间 t_r 为 1.59 s ,峰值时间 t_p 为 1.8 s ,调节时间 t_s 为 2.8 s ,超调量 σ_p 为 14% 稳态误差为 0. 表明所设计的控制系统具有良好的动态性能和稳

态性能.

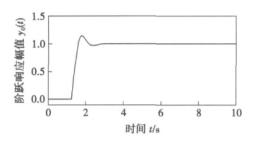


图 7 控制系统模型的单位阶跃响应曲线

Fig. 7 Unit step response curve of control system model

3 结 论

- (1) 采用阶跃响应法确定出了钛合金电子束焊电子束流与熔宽的数学模型的类型和结构,并且计算出了该模型的参数,建立了电子束流与熔宽的传递函数模型.
- (2) 利用获得的电子束流与熔宽的数学模型, 采用 Z-N 经验公式法确定出了 PID 控制器参数 在此基础上建立了钛合金电子束焊熔宽 PID 控制系统模型.
- (3) 用建立的模型进行了仿真试验,试验结果表明所设计的控制系统具有良好的动态性能和稳态性能.

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MAIN TOPICS ABSTRACTS & KEY WORDS

Simulation of plasma deposition manufacturing process for functionally graded material using level-set approach

WANG Guilan¹, WANG Qing¹, WANG Xiangping², ZHANG Haiou²(1. State Key Lab of Materials Processing and Die & Mould Technology, Huazhong University of Science & Technology, Wuhan 430074, China; 2. State Key Lab of Digital Manufacturing Equipment & Technology, Huazhong University of Science & Technology, Wuhan 430074, China). pp 1–4, 8

Abstract: A multiphase continuum numerical model based on level-set approach and finite volume method (FVM) was proposed to simulate plasma deposition manufacturing (PDM) process for functionally graded material (FGM) . Based on this model , molten pool behavior , temperature field and flow field as well as solute diffusion subjected to a variety of heat power were investigated. The simulation results manifest that heat power has significant influence on the composition distribution and the final property of FGM. An experiment in Al $_2$ O $_3$ -stainless steel FGM deposition verified the validity and feasibility of this simulation.

Key words: plasma deposition manufacturing; functionally graded material; multiphase continuum model

Modeling and simulation of PID control system of titanium alloy electron beam welding SHI Mingxiao¹, ZHANG Binggang², CHEN Guoqing², FAN Ding¹(1. State Key Laboratory of Gansu Advanced Non-ferrous Metal Materials, Lanzhou University of Technology, Lanzhou 730050, China; 2. State Key Laboratory of Advanced Welding and Joining, Harbin Institute of Technology, Harbin 150001, China). pp 5–8

Abstract: Due to the complexity of the welding process, it was difficult to obtain the analytic mathematical model of dynamic process of molten pool, that was why the modeling of welding process became a major problem to the welding process control. For the purpose of establishing the model of dynamic process of molten pool of the electron beam welding. The step-response method was used to determine the type and structure of mathematical model of electron beam current and molten width of titanium alloy electron beam welding, the parameters of model were calculated , and the transfer function model of electron beam current and molten width was established. The Z-N empirical formula was used to obtain the parameters of PID controller based on the mathematical model. On this basis the MATLAB/SIMU-LINK platform was used to establish the model of PID control system of molten width of titaniumalloy electron beam welding. The simulation experiment was done and the results show that the dynamic performance and steady-state performance of designed control system are eminent.

Key words: electron beam welding; molten width; modeling; simulation

Effect of joint clearances on microstructure and property of brazed SiC ceramics LI Zhuoran¹, XU Xiaolong¹, LIU Wenbo², LU Zhiguo² (1. State Key Laboratory of Advanced Welding and Joining, Harbin Institute of Technology, Harbin 150001, China; 2. AVIC Harbin Dongan Engine Co., Ltd., Harbin 150066, China). pp 9–12

Abstract: The effect of joint clearance on microstructures and property of SiC ceramics brazed with Ti-Zr-Ni-Cu brazing filler was studied by means of SEM ,EDS and XRD. The joint microstructure from SiC ceramics to the center of brazing seam are TiC , Zr(s,s) , $Ti_5Si_3 + Zr_2Si$, $Ti(s,s) + Ti_2(Cu,Ni)$ and (Ti Zr) (Ni ,Cu) successively. When the joint clearance is 30 $\mu m \sim 50 \ \mu m$, Si reacts with Ti and Zr element , and produces a small quantity of tiny needlelike silicon compounds. Homogeneous solid solution is full of the brazed joint. The mechanical property is superior and the shear strength can reach 117 MPa. When the joint clearance is less than 30 µm, longstring silicon compounds are generated, and run through the brazing beam. Meanwhile, successive and thick TiC layer is produced. Both of these two factors lower the mechanical property. When the joint clearance is more than $50~\mu m$, Ti-based intermetallic increases and eutectic compounds are produced, which reduce the mechanical property.

Key words: SiC ceramics; brazing; microstructure; shear strength

High temperature wear behavior of WC particles reinforced Ni-based plasma-sprayed coating by laser remelting

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Abstract: To improve the properties of tungsten carbide (WC) particles reinforced Ni-based coating prepared by plasma spraying, laser remelting experiment was carried out and the effects of laser remelting on microstructure and properties on coating were studied. The surface morphology, microstructure, phase and microhardness of coatings were investigated by using scanning electron microscopy (SEM), X-ray diffractometry and microhardness measurement. Meanwhile, the high temperature fiction and wear behavior for both coatings were comparatively studied. The results showed that the defects of as-sprayed coating like lamellar stacking microstructure and pores were eliminated by laser remelting, and the remelted coating possessed a denser microstructure. In addition, WC particles partly dissociated at the edge, and the dendrite grew from the surface of the partially melted WC particles under high energy density laser. The laser-