

# 热塑挤压改善镁合金熔焊接头疲劳性能及其微观机制

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**摘 要:**以 AZ31B 镁合金为试验材料, 用同质成分丝材作填充材料进行 TIG 对接焊。将获得的熔焊接头在真空热压炉中实施 400 °C 恒温热塑挤压变形, 然后再进行室温疲劳性能试验。对试验试样借助于扫描电镜、微观金相及能谱分析等手段探讨热塑挤压改善镁合金熔焊接头疲劳性能方法及其微观机制。结果表明, 热塑挤压能够有效地提高镁合金熔焊接头疲劳寿命, 其改善机制主要基于 400 °C 恒温挤压变形有助于熔合区氧化物夹杂的破碎及微裂纹的密实、愈合, 并促使熔合区粗大  $\alpha$  晶粒形成亚晶界以细化组织; 同时热塑挤压变形又使焊缝组织通过动态再结晶实现组织重构, 并改变  $\beta$  相沿晶界网状分布为重溶后在  $\alpha$  相晶内呈弥散分布。

**关键词:** 热塑挤压; 镁合金; 熔焊接头; 疲劳性能

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## 0 序 言

镁合金熔焊接头的力学性能是决定其能否应用于重要构件制造的关键, 但金属镁系室温单一滑移系的密排六方晶体结构, 具有低塑性本征特性。对于 Mg-Al 系的镁合金, 焊缝凝固后, 铸态组织中强化相在非平衡冷却条件下往往以不连续析出方式于  $\alpha$ -Mg 固溶体晶界析出。这造成镁合金熔焊接头强度、塑性直至疲劳性能均存在较大的问题, 难于实现与镁合金母材的良好匹配<sup>[1-3]</sup>。对于变形镁合金, 该问题更显突出。镁合金熔焊接头疲劳特性的改善可着眼于强韧化控制。从理论上分析, 可以利用加热熔焊接头至一定温度诱发开通密排六方棱柱面及锥面新滑移系, 在具有一定塑性状态下实施热塑挤压, 使其发生动态回复与再结晶并引导强化相固溶后重新以连续析出方式在  $\alpha$ -Mg 固溶体晶内弥散析出, 同时通过细晶强化达到改善疲劳特性的效果<sup>[4-7]</sup>。基于这一思想, 文中以 AZ31B 镁合金钨极氩弧焊获得的熔焊接头为例, 采用真空热塑挤压方法进行工艺及疲劳性能试验, 并借助于微观分析手段对其机理作深入分析与讨论, 为镁合金熔焊接头疲劳性能的改善探索出一条有效途径。

## 1 试验方法

试验所用材料为 AZ31B 变形镁合金, 合金元素成分(质量分数)为 2.5% ~ 3.5% Al, 0.5% ~ 1.5% Zn, 0.2% ~ 0.5% Mn, 焊丝为与 AZ31B 同质成分型材经挤压、拉拔成直径为 2 mm 丝材。试验中镁合金试板尺寸由两块 200 mm × 60 mm × 4.6 mm 组成对接焊试板, 开 60°V 形坡口, 焊前用有机溶剂清洗焊接区域油污并用砂纸打磨去除氧化膜。用手工钨极氩弧焊进行双面焊, 正面施焊 2 道, 背面施焊 1 道。焊接工艺参数电流为 110 ~ 120 A, 电弧电压为 21 ~ 23 V, 焊接速度为 8 ~ 11 mm/s。正面焊缝余高控制在 2 mm, 反面焊缝余高控制在 1 mm。焊后沿垂直焊缝方向以焊缝为中心截取若干宽度为 24 mm 的条形试块。将截取出的一部分试块按图 1 所示放置在真空热压炉内利用焊缝余高对接头进行周期性热塑挤压, 挤压工艺参数见表 1。经热塑挤压后焊缝与母材完全齐平, 将未经热塑挤压的另一部分焊态试块焊缝打磨平整, 然后将热塑挤压态和焊态两种试块按国家标准 GB-T13816—1992《焊接接头脉动拉伸疲劳试验方法》加工成疲劳试样, 分别在德国 Zwick/Reoll 公司进口 AMSLER100HFP5100 高频疲劳试验机上进行疲劳试验(应力比  $r = 0.1$ , 加载频率  $f = 70 \sim 80$  Hz)。疲劳断口用 JSM-6360LV 扫描电镜及其附带能谱分析仪进行分析。焊接接头微观组织用金相显微镜观察。

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表 1 热塑挤压工艺参数

Table 1 Parameters of thermo plastic extrusion

挤压时间 $t/\text{min}$	恒温温度 $T/^\circ\text{C}$	冷却速率 $\omega/(^\circ\text{C}\cdot\text{min}^{-1})$	加压压力 $F/\text{N}$	压缩量 $d/\text{mm}$	加压次数 $n$	真空度 $k/\text{mPa}$
30	400	6	5 000	4	3	7.3



图 1 试样在真空室内挤压照片

Fig. 1 Photo of extruding specimen inside vacuum room

## 2 结果分析与讨论

### 2.1 热塑挤压对熔合区裂纹扩展的止裂延寿效应

表 2 为焊态与在真空热压炉中经  $400^\circ\text{C}$  , 30 min 热挤压态两种状态的疲劳试验结果.

表 2 疲劳试验结果

Table 2 Results of fatigue tests

试样 状态	试样 编号	名义应力范围 $\Delta\sigma/\text{MPa}$	失效循环数 $N_f/(10^3 \text{ 周次})$	断裂位置
焊态	1-1	90	3.229	熔合区
	1-2	70	11.533	
	1-3	60	16.311	
热挤 压态	2-1	90	63.217	熔合区
	2-2	70	83.297	
	2-3	60	94.375	

试验结果表明,在相同疲劳加载条件下,经  $400^\circ\text{C} \times 30 \text{ min}$  的热挤压动态塑性变形后, AZ31B 镁合金熔焊接头断裂前的循环次数得到有效地增大,但断裂位置及裂纹走向几乎没有什么变化. 为探讨热塑挤压对阻止熔焊接头疲劳裂纹扩展的有益作用,对经焊态和热挤压态的焊接接头用扫描电镜观察发现,两者熔合线均分布有若干点状物构成的白色带(图 2). 高倍 SEM 显示该白色带为夹杂物引起的微裂纹(图 3a),但经热塑挤压后微裂纹在尺寸上有变细小的趋势. 能谱分析表明夹杂物的主要成分为镁、铝、氧,与其它部位相比存在着较高的氧含量

(图 3b). 据此可以初步断定是  $\text{MgO}$  引起的夹杂物. 该氧化物最大的可能来源是焊前母材表面及坡口未清除干净的残余氧化膜带入焊接区域而形成的. 由于其熔点高达  $2\,500^\circ\text{C}$  以上,在凝固时极易呈粒状以夹杂物形式残留于熔合区进而诱发微裂缝. 疲劳裂纹源一旦在试样表面应力集中部位形成,便很容易沿熔合区的这些夹杂物进行扩展. 但经热挤压变形处理后,一方面夹杂物因受力而得到破碎;另一方

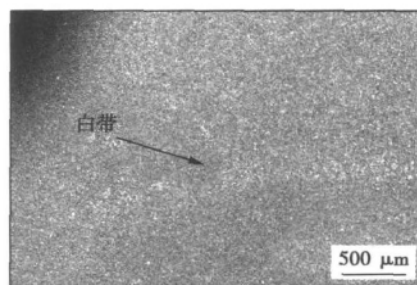
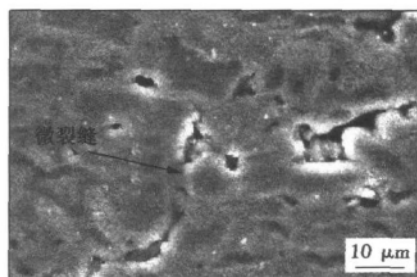
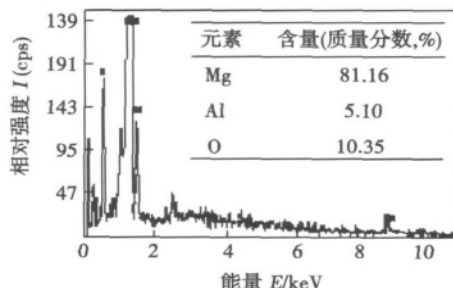


图 2 熔合区白带 SEM 形貌

Fig. 2 SEM photo of white belt in fusion zone



(a) 熔合区微裂纹



(b) 夹杂物能谱

图 3 熔合区夹杂物引发微裂缝及夹杂物微区能谱图  
Fig. 3 Micro-crack resulted by inclusion in fusion zone and micro-area energy spectrum of inclusion

面夹杂物引起的微裂缝在一定程度上受挤压而变得密实,甚至得到愈合。这无疑对疲劳裂纹地扩展起到有效的缓冲作用,因而是疲劳延寿的一个重要原因。

## 2.2 热塑挤压对疲劳断裂机制的影响

图 4 为名义应力范围  $\Delta\sigma = 70$  MPa 载荷下的 TIG 焊接头在焊态和经热塑挤压态两种条件下疲劳试样断口宏观及微观形貌。从断裂机理来看,二者断裂均源于试样棱角尖端的应力集中处,裂纹走向沿对角线向试样中心扩展(图 4a, b)。焊态疲劳断口找不到十分明显疲劳辉纹,而经热塑挤压后的疲劳辉纹较为明显,且辉纹十分密集(图 4b)。这表明焊态下疲劳裂纹的扩展区很短,很快即进入失稳扩展的瞬断区。但经热塑挤压后裂纹扩展区得到有效地扩张,即经历了一定的扩展时期后才转入失稳扩展期,且细密的疲劳辉纹表征裂纹扩展速率明显减小,这是疲劳寿命得到延长的主要表现。疲劳断口 SEM 分析还可看出,焊态主要呈现解理断裂特征,断口中有较多舌状花样,时常伴随有二次裂纹的产生(图 4c)。相比之下,热塑挤压态的疲劳断口显示晶粒发生了较大的变形,沿某一方向晶粒已被拉长,有一定的撕裂迹象,因此可以说明断裂前伴随有一定的塑性变形发生。

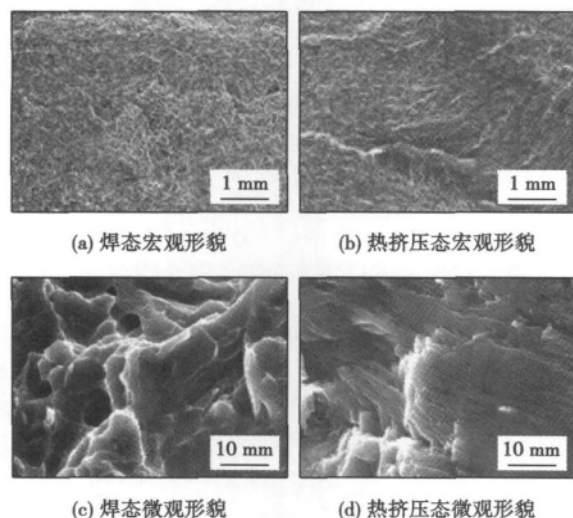


图 4 疲劳断口 SEM 形貌

Fig. 4 SEM morphology of fatigue fracture

镁合金熔焊接头熔合区组织仍然是  $\alpha$ -Mg 固溶体基体上分布着  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 析出相。焊态下的熔合区组织由于  $\alpha$ -Mg 相在焊接热循环作用下没有发生固态二次相变,因此  $\alpha$ -Mg 相极易长大成为粗晶组织。而  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 析出时又大都沿平行于密排六方

基面的惯习面析出,但基面自身又是镁合金室温发生变形时唯一滑移面。因此焊态下脆性  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 相的存在很可能导致此处塑性严重降低,断口主要表现为脆性断裂特征<sup>[7,8]</sup>。但经 400 °C 热塑挤压变形时,一方面熔点处于该温度的残留  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 相此时将发生熔化;另一方面该温度发生塑性变形已完全能够开通密排六方晶体棱面及锥面新滑移系,使得晶体内部滑移更加容易进行。这将促使亚晶界的形成,最终导致晶粒细化,室温塑性得到改善,疲劳裂纹扩展时需要更多的能量克服塑性变形,因而断口将表现为有规则的周期性疲劳辉纹。

## 2.3 热塑挤压对熔焊接头微观组织重构的影响

热塑挤压改善 TIG 焊接头疲劳性能的另一重要方面是对微观组织的改性作用。图 5 为熔焊接头微观金相形貌。金相显微分析表明焊态下焊缝金属微观形貌为等轴树枝晶(图 5a)。对于共晶型的 Mg-Al 系镁合金,在偏离平衡的凝固条件下溶质元素 Al 在枝晶间的偏析严重,其结果是造成晶界铝的富集,随后的非平衡冷却过程中在晶界形成较多的  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 脆性相。这一方面造成晶界弱化,另一方面又使得晶内铝相对贫乏进而降低固溶强化效果。与此同时在熔合区附近,由于焊接热循环作用下  $\alpha$ -Mg 固溶体不发生二次相变,因而晶粒长大现象十分严重(图 5b)。这些都造成焊接接头强韧性低下,疲劳寿命较短。分析断口附近的金相组织正好对应于熔合区的粗晶部位(图 5c),因此裂纹启裂及扩展大都沿熔合区进行。当焊接接头在 225 °C 以上温度进行热塑挤压时,密排六方结构的镁合金此时可以开动棱柱面新滑移系,塑性得到有效改善。晶内滑移

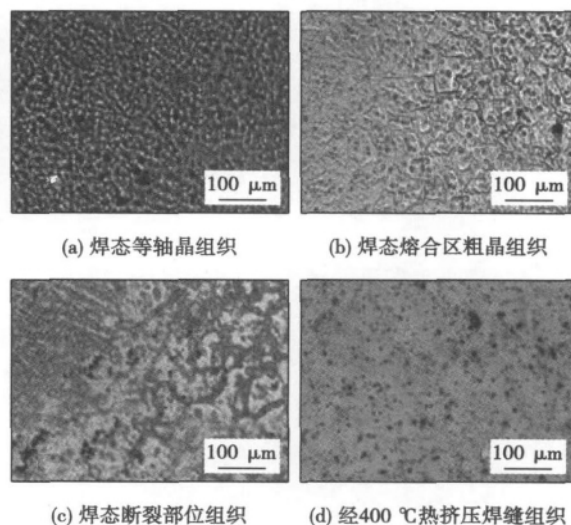


图 5 熔焊接头微观金相形貌

Fig. 5 Microstructure of fusion weld joint

往往使位错在晶界塞积从而积蓄较多的畸变能,动态再结晶极易充分进行而促使先前的枝晶得以改组重构<sup>[9,10]</sup>。同时对 AZ31B 镁合金而言,合金元素 Al 含量不高,一般在 3.0% 左右,而在共晶温度(436℃)附近,铝在镁中的溶解度可高达 12.6%。因此原先富集在晶界的  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 相此时很容易重新固溶于  $\alpha$  相中,在随后的近平衡冷却过程以弥散质点形式主要在晶内以连续析出方式析出,极大地克服了原先存在于晶界的脆化作用并在晶内发挥弥散强化作用(图 5d)。因而强度及塑性均得到一定的提高,这对阻止疲劳裂纹扩展,进而改善疲劳特性也起到十分有益的作用。

### 3 结 论

(1) 对 AZ31B 镁合金熔焊接头进行适当的热塑挤压变形,可在一定程度破碎熔合区残留氧化夹杂物,并使得夹杂物引发的微裂缝受挤压而密实、甚至愈合,从而有效地缓冲疲劳裂纹的扩展速率。

(2) 热塑性挤压可使镁合金熔焊接头熔合区粗晶组织通过塑性变形形成亚晶界而细化,并抑制  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 沿平行于基面惯析面的析出,使裂纹扩展具有细密疲劳辉纹特征,从而延长室温疲劳裂纹的稳定扩展期。

(3) AZ31B 镁合金熔焊接头经 400℃×30 min 热塑挤压变形,铸态焊缝经过再结晶实现组织重构,有效地消除了枝晶偏析及  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 相沿晶界分布对韧性的不利影响,通过细晶强化及  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> 相的重溶并随后在晶内弥散析出强化,对疲劳强度的改善起到十分积极的作用。

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## MAIN TOPICS ,ABSTRACTS & KEY WORDS

### Improving fatigue properties with thermo plastic extrusion and micro-mechanism for fusion welded joint of magnesium alloy

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**Abstract:** As an example of experimental material , AZ31B magnesium alloy was butt welded with filling wire of similar composition by TIG welding method. The fusion welded joints attained were put in a vacuum furnace to be thermo plastically extruded at constant temperature of 400 °C and then the fatigue properties of joints were tested at room temperature. With the help of scanning electronic microscopy , optical metallographic microscope and energy spectrum analyzer etc. , the experimental specimen were analyzed to explore the way and micro-mechanism of improving fatigue properties with thermo plastic extrusion for fusion welded joint of magnesium alloy. The results show that with thermo plastic extrusion , the fusion welded joint fatigue life of magnesium alloy can be improved effectively , this mechanism is chiefly due to beneficial effecting for oxide to be broken and micro-crack to be closed or healed with constant temperature extrusion at 400 °C , also the coarse  $\alpha$  grain in fusion zone may be promoted to transform into sub grain boundary , therefore a refining microstructure can be attained. Meanwhile it causes weld microstructure to reconstruct by way of dynamic re-crystallization and changing network-like distribution of  $\beta$ -Mg<sub>17</sub>Al<sub>12</sub> precipitate on  $\alpha$ -Mg grain boundary of weld microstructure and making it solid solve to re-precipitate in way of dispersed distribution inside  $\alpha$  grain.

**Key words:** thermo plastic extrusion; magnesium alloy; fusion welded joint; fatigue properties

### Microstructures and electrochemical performance of 6082-T6 aluminum alloy welds prepared by bobbin friction stir welding

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**Abstract:** Friction-stir welding ( FSW ) of 12-mm-thick plates of 6082 Al was conducted by bobbin-tool. Microstructure and potentiodynamic polarization behavior were examined by optical microscope and CHI660A three-electrode electrochemical workstation. The results indicated that the microstructure of the weld was remarkably reshaped under the condition of the stir-pin rotation speed 600 r/min , travel speed 300 mm/min and tilt angle 0° of the stirring tool. The equiaxed grain microstructures were formed due to dynamical recrystallization in the nugget zone. The corrosion test results showed that the pitting corrosion

of base material was more severe than that of bobbin-tool FSW weld , and the corrosion morphology was much rough than that of bobbin-tool FSW weld. The corrosion potential of the weld was positive and higher than that of parent material. The corrosion current density of bobbin-tool FSW weld was lower than that of parent material.

**Key words:** 6082 aluminum alloy; friction stir welding; microstructures; electrochemical corrosion

### Design of oxygen nozzle for oxy-propane gas cutting

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**Abstract:** According to the characteristics of oxy-propane gas cutting nozzle and the aerodynamic principles a software has been developed for oxygen nozzle design. Oxygen nozzle flow profile data and other parameters of oxygen flow would be calculated out by inputting several processing parameters. The paper presented an example of oxygen nozzle designed by the software tool. Computational fluid dynamics ( CFD ) simulation is employed to analyze the flow field characteristic parameters as the oxygen nozzle is working under the design pressure. Analysis of the distribution of oxygen jet flow velocity , mach number , and oxygen mass fraction have been carried out. The results show that the designed oxygen nozzle works and also indicate the correctness and reliability of the software.

**Key words:** gas cutting; large thickness; cutting nozzle; oxygen nozzle; propane; numerical simulation

### Numerical analysis to study effect of turn number on residual stress and deformation of butt welded 20MnMoNb super-thick tube-sheet

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**Abstract:** The butt-welding process of 390mm 20MnMoNb super-thick plates for manufacturing the tube-sheet of a large scale ethylene oxide reactor was simulated by finite element program-ABAQUS. The residual stress and deformation in the welded tube-sheet are obtained. The welding groove was double U-shape and it was welded on both sides by turning over the tube-sheet continuously with a travelling crane. The effect of the turning numbers on the residual stress and deformation was investigated. The results show that angular deformation occurred in the tube-sheet and large tensile residual stresses is generated in the weld and the heat affect zone closed to the external surfaces of the tube-sheet , while compressive residual stress inside the