

Ni+Cu 为中间层的 TC4 与 ZQSn10-10 的扩散连接试验分析

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摘 要: 对钛合金 TC4 与铜合金 ZQSn10-10 异种金属扩散连接进行了试验研究。采用 Ni+Cu 中间层时, 其最佳工艺参数为连接温度 850 ℃, 连接时间 20 min, 连接比压力 10 MPa, 抗拉强度达到 155.8 MPa。在 TC4/Ni 界面上, 形成了成分逐渐变化的互扩散层。经微区成分分析可知, 连接温度为 800 ℃时, 界面主要为 Ni_3Ti ; 850 ℃时, 界面主要为 Ni_3Ti 及 NiTi ; 880 ℃时, 界面主要为 NiTi_2 , Ni_3Ti 及 NiTi 。通过 EDS 和 XRD 分析, 接头强度主要取决于镍钛金属间化合物的种类及厚度。

关键词: 钛合金; 锡青铜; 扩散连接; Ni+Cu 中间层; 金属间化合物

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0 序 言

钛及钛合金具有高的比强度、优良的耐热、耐腐蚀性及断裂韧度等优点, 在航空航天、石油化工、医疗及地质等领域受到极大青睐^[1]。锡青铜具有较好的抗腐蚀性能和较高的耐磨润滑性能, 某航空发动机中精密摩擦使用的是锡青铜与钢的复合材料结构^{2,3}, 如果采用锡青铜与钛合金的复合材料结构, 将会大大减轻其重量, 提高结构效率。因此, 如何实现锡青铜与钛合金的优质连接是国内外普遍关注而又亟待解决的问题。

众所周知, 扩散连接技术是一种精密的固相连接方法⁴, 特别适合于性能差别大、互不溶解、相互间易产生脆性相的异种材料连接⁵。但应注意解决由于两种金属之间的物理、化学性能差异带来的一些问题, 如通过添加中间层来减缓残余应力、限制有害元素的金属间化合物的产生等, 以提高连接强度⁶⁻⁸。为此, 作者对 Ni+Cu 为中间层时钛合金与铜合金扩散连接接头的组织和性能进行了试验研究。

1 试验材料及方法

试验采用的钛合金为 TC4, 铜合金为 ZQSn10-

10。其化学成分及主要性能如表 1 所示。金相试样和拉伸试样采用的尺寸分别为 $\phi 16 \text{ mm} \times 5 \text{ mm}$ 和 $\phi 16 \text{ mm} \times 30 \text{ mm}$ 。中间层采用镍箔 ($50 \mu\text{m}$) 和铜箔 ($50 \mu\text{m}$), 组成 TC4/Ni/Cu/ZQSn10-10 试样。

连接前试样被连接表面逐次用 320 号、500 号、1000 号、1500 号金相砂纸磨光, 连接面用氢氟酸和硝酸混合溶液去除钛合金表面的氧化膜; 用硫酸溶液去除铜合金表面的氧化膜; 用丙酮去脂。中间层镍箔和铜箔也逐次用 500 号、1000 号、2000 号金相砂纸磨光, 然后用丙酮或无水酒精擦拭即可。

连接试验是在辐射加热扩散炉中进行, 真空度为 $1.33 \times 10^{-3} \text{ Pa}$ 。采用美国的 JSM6700F 冷场发射电镜对连接接头横截面进行组织分析, 并用其自带的能谱仪 (EDS) 进行线扫描和点的成分检测来分析元素扩散情况。用拉伸试验来评价接头的强度。

2 试验结果及分析

2.1 连接温度对接头性能的影响

连接温度 T 是最重要的工艺参数, 对接头性能的影响见图 1。当 $T=800 \text{ }^\circ\text{C}$ 时, 接头的强度很低, 仅为 49.8 MPa。拉伸试验时, 断裂沿 TC4/Ni 连接面发生, 断口中存在大量的未接合区 (图 2)。随着 T 升高, 接头强度逐渐增加, 当 $T=850 \text{ }^\circ\text{C}$ 时, 接头强度达到最高值 155.8 MPa。当 T 继续升高时, 接头强度急剧下降。

表 1 TC4 和 ZQSn10-10 主要化学成分及力学性能

Table 1 Chemical composition and mechanical propriety of TC4 and ZQSn10-10

材 料	化学成分(质量分数 %)										力学性能		
	Ti	Al	V	Fe	O	Cu	Sn	Pb	Ni	P	抗拉强度 R_m /MPa	屈服强度 R_{el} /MPa	断后伸长率 A (%)
TC4	余量	6.2 ~ 6.4	4.0	0.08	0.16	—	—	—	—	—	900~1 100	810	17
ZQSn10-10	—	—	—	—	—	余量	9.2	7.8	0.025	0.13	240	125	28

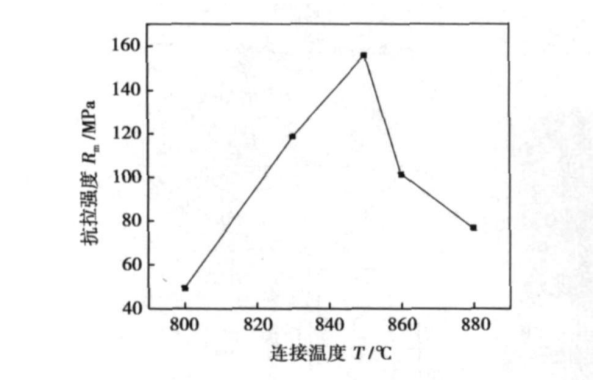


图 1 连接温度对接头性能的影响($p=10\text{ MPa}$ $t=20\text{ min}$)
Fig 1 Influence of temperature on mechanical property of joint

连接强度从本质上讲是由界面组织结构决定的。当连接温度较低时,材料的屈服强度较高,界面处只是局部微观突起优先接触而发生塑性变形,有效接触面积小,原子热激活不够,形不成完全的物理接触,不能形成可靠的连接;随着连接温度的升高,母材的屈服强度下降,界面处塑性变形增加,接头的有效接触面积不断增大,原子的扩散更加充分,界面反应进一步进行。此时,由于形成的金属间化合物层很薄且呈弥散分布,对强度的危害较小,因此在 850 °C 时的强度达到最大值。随着连接温度的再进一步升高,虽然元素的扩散及反应更加充分,却使连接界面金属间化合物种类增多、层厚加大。同时,又

由于连接温度过高导致锡青铜中 Sn、Pb 等合金元素的熔化、烧蚀和偏析,进一步弱化了接头性能,使接头强度开始迅速下降。

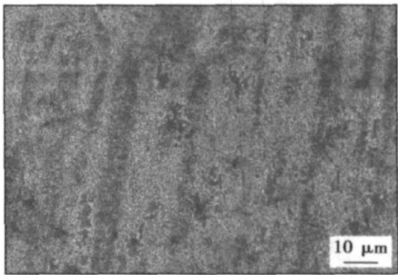


图 2 TC4/ Ni 界面断口形貌($T=800\text{ °C}$, $p=10\text{ MPa}$ $t=20\text{ min}$)
Fig 2 Fractograph of TC4/ Ni interlayer

扩散连接接头金相显微形貌可分为五个区域见图 3a,依次为 A 区——铜合金 ZQSn10-10 母材;B 区——ZQSn10-10/ Cu 界面和中间层 Cu 的一部分;C 区——中间层 Cu/ Ni 互溶区;D 区——Ni/ TC4 界面和中间层 Ni 一部分;E 区——钛合金 TC4 母材。接头元素线扫描分析见图 3b,中间层 Cu 向其两侧扩散, B 区中的 Cu/ ZQSn10—10 界面上的 Cu 原子沿晶界扩散,且扩散速度很快,因此两者之间的界限基本消失。在 Cu, Ni 互溶的 C 区,由于 Ni 与 Cu 同属

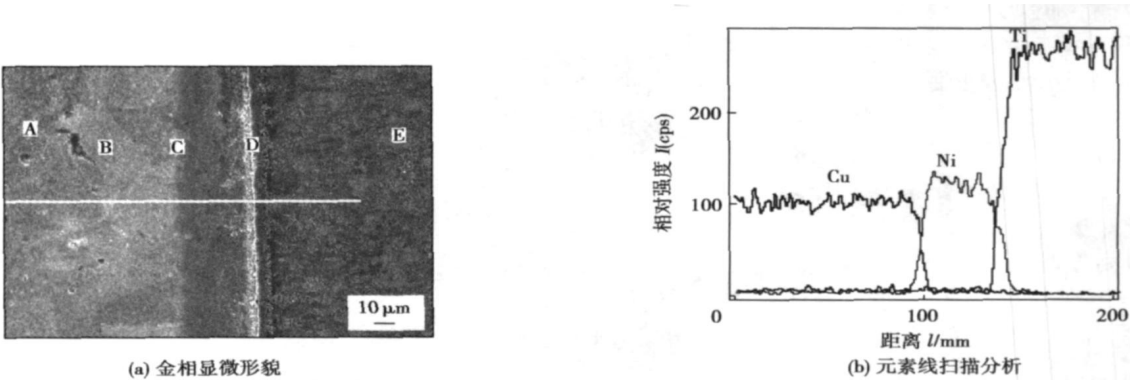


图 3 采用 Ni+ Cu 中间层时 TC4 与 ZQSn10—10 扩散连接接头($T=850\text{ °C}$, $p=10\text{ MPa}$, $t=20\text{ min}$)
Fig 3 SEM micrograph and element line scanning curve of TC4 and ZQSn10-10 with Ni+ Cu interlayer

于面心立方晶格,形成无限固溶体而不形成金属间化合物,镍与铜中间层扩散比较充分,过渡均匀。中间层Ni向其两侧充分扩散,尤其是Ni与Ti之间发生了较强烈的相互扩散,而且元素分布曲线上出现了平台,说明在扩散层中形成了镍钛金属间化合物薄层。如图3a中D区,在钛合金TC4/Ni界面上,由于两侧金属硬度、密度等相差较大,接合界面呈明显带状。经腐蚀后在界面上可见异于两侧金属的过渡层组织,即金属间化合物层。

2.2 连接时间和连接压力对接头性能的影响

连接时间并不是一个完全独立的参数,应根据连接温度和压力合理选择。在 $T=850\text{ }^{\circ}\text{C}$, $p=10\text{ MPa}$ 的条件下,研究了连接时间对接头强度的影响,试验结果见图4。在其它参数不变的情况下,适当延长连接时间,可使扩散进行得比较充分,有利于接头成分和组织的均匀,如图4中当 $t=20\text{ min}$ 时,强度达到最高155.8 MPa。但时间过长,即使接头的抗拉强度不低,但由于扩散连接温度一般应为母材熔点的50%~80%(处于再结晶温度以上),这样容易造成接头区晶粒粗大,试件会产生不允许的变形,接头性能下降。

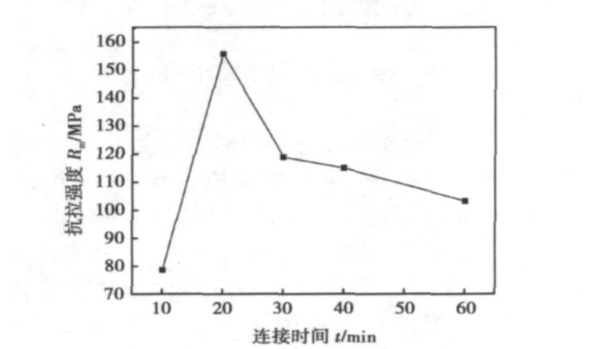


图4 连接时间对接头性能的影响 ($T=850\text{ }^{\circ}\text{C}$, $p=10\text{ MPa}$)
Fig. 4 Influence of bonding time on mechanical property of joint

试验中,连接比压力的主要作用是使连接表面微观凸起部分产生塑性变形,使之达到紧密接触状态,并提供变形能为原子的扩散创造条件。依据相关文献[9],文中的连接压力选择为10 MPa。

为了进一步探明试验条件下中间生成相的情况,必须对接头进行微区成分分析。

2.3 中间生成相的分析

由于拉伸试件都从Ni/TC4界面断裂,为了确定金属间化合物区的相组成,对应Ni/TC4断口的X射线衍射谱,见图5。结合能谱扫描,可以看到, $T=800\text{ }^{\circ}\text{C}$ 时,界面主要为 Ni_3Ti (图5a); $T=850\text{ }^{\circ}\text{C}$ 时,界面主要为 Ni_3Ti 及 NiTi 相(图5b); $T=880\text{ }^{\circ}\text{C}$ 时,界

面主要为 NiTi_2 , Ni_3Ti 及 NiTi (图5c)。

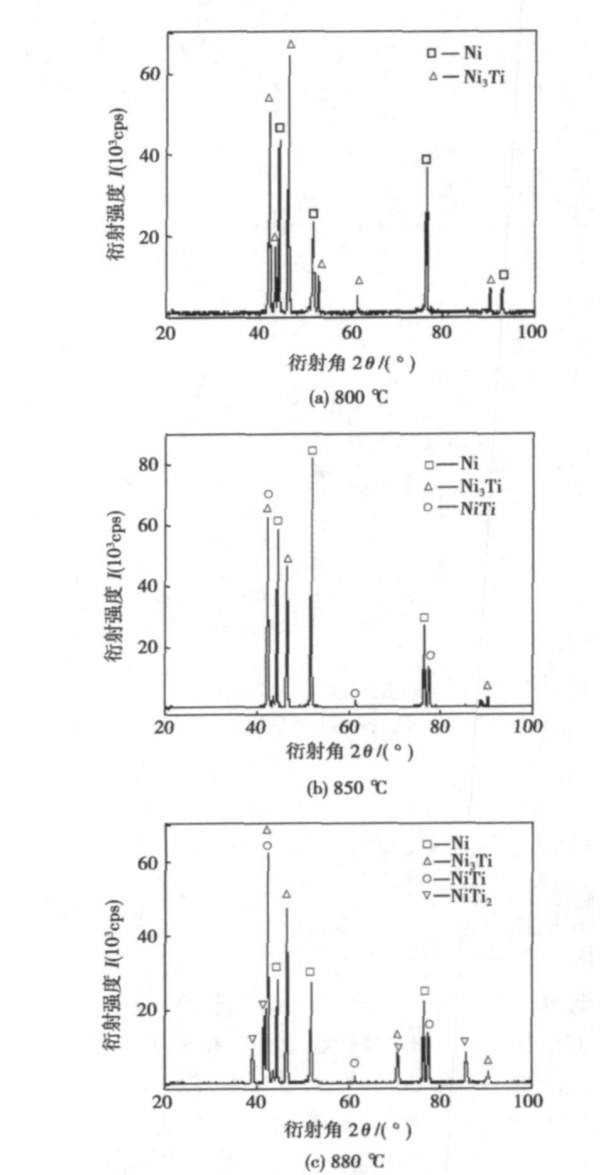


图5 Ni/TC4 界面断口 XRD 分析
Fig. 5 XRD patterns of fracture surface of Ni and TC4

由于镍在钛中的扩散系数远远大于钛在镍中的扩散系数,因此其反应过渡层主要是由镍向钛扩散形成。文献[10]给出了各反应相的生成能与温度之间的关系为

$$\Delta G(\text{NiTi}_2) = -49\,120 + 17.208T \text{ (J/mol)}, \quad (1)$$

$$\Delta G(\text{NiTi}) = -54\,600 + 18.133T \text{ (J/mol)}, \quad (2)$$

$$\Delta G(\text{Ni}_3\text{Ti}) = -55\,585 + 15.962T \text{ (J/mol)}, \quad (3)$$

式中: T 为绝对温度。

由式(1)~式(3)可以看出,在连接温度低于1173 K(即900 °C)时,镍与钛之间最易产生 Ni_3Ti ,其次是 NiTi ,最不容易产生 NiTi_2 。这与文中试验结

果相一致。因此, 接头力学性能主要受镍钛之间形成的金属间化合物层的影响。

3 结 论

(1) 采用 Ni+Cu 中间层扩散连接钛合金 TC4 与铜合金 ZQSn10—10, 其最佳工艺参数是连接温度 850 ℃, 连接时间 20 min, 连接比压力 10 MPa, 抗拉强度达到 155.8 MPa。

(2) 在钛合金 TC4/Ni 界面上, 形成了成分逐渐变化的互扩散层, 即金属间化合物区。经微区成分分析, 连接温度为 800 ℃时, 界面主要为 Ni₃Ti; 850 ℃时, 界面主要为 Ni₃Ti 及 NiTi; 880 ℃时, 界面主要为 NiTi₂, Ni₃Ti 及 NiTi。

(3) 通过 EDS 和 XRD 分析, 接头强度主要取决于镍钛金属间化合物的种类及厚度。

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Diffusion bonding of TC4 to ZQSn10—10 with nickel and copper interlayers

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Abstract: The experimental investigation of the diffusion bonding of TC4 to ZQSn10—10 in vacuum was carried out by using pure nickel and copper as the transition metal. Experimental results show that the optimum bonding parameters were selected as follows: bonding temperature 850 °C, bonding time 20 min and bonding pressure 10 MPa. So the strength of the joint without obvious shape changing was up to 155.8 MPa, which the strength can reach to about 65% of the strength of base metal ZQSn10—10. Various reaction layers appeared in TC4 and Ni interface. When the rest condition was same, Ni₃Ti phase was produced at the interface zone at 800 °C; and there appeared Ni₃Ti and NiTi phases at 850 °C; and there produced NiTi₂, Ni₃Ti and NiTi phases at 880 °C. The strength of the joint lied on types and thickness of nickel and titanium inter-metallic compounds.

Key words: titanium alloy; tin-bronze; diffusion bonding; nickel and copper interlayers; intermetallic compound

Prediction system of CTOD for high strength pipeline steel welded joint based on back propagation artificial neural network

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Abstract: Aiming at limitation of selecting the main technical parameters for high strength pipeline steel welding in practical operation, a back propagation artificial neural network (ANN) was established with Visual C++ 6.0 for predicting the welding performance parameter crack tip opening displacement (CTOD) of high strength pipeline steel joint. Based on the experiment data, the average heat input, wall thick, preheat temperature and joint region were used as the input parameters of ANN, which includes one input layer with 4 nodes, one hidden layer with 14 nodes, and Sigmoid activation function. The average absolute error of prediction result is 15.4%. The number of the sample whose error is less than $\pm 20\%$ is about 93.3% in total 15 experimental data. The result showed that ANN method is a kind of effective method to predict the welding performance parameter CTOD of high strength pipeline steel welded joint. The ANN system can be used as selecting and optimizing the key welding parameters.

Key words: artificial neural network; welded joint; crack tip opening displacement; prediction; performance parameter

Effects of welding residual stress on creep of welded joint

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Abstract: By the finite element analysis codes ABAQUS and

the function of RESTART, a sequential coupling calculating program for creep based on welding residual stress has been developed by establishing the model of temperature field, residual field and creep analysis. The welding residual stress and creep of welded joint for Cr5Mo furnace tubes was simulated by this program. This method established the basis for calculating the effects of welding residual stress on the creep of welded joint. A creep comparison between the welded joint tubes with welding residual stress and with internal pressure was made. The results showed that the tube creep was decided by the welding residual stress, although the welding residual stress was relaxed in a short time.

Key words: Cr5Mo furnace tubes; welded joint; welding residual stress; stress relaxation; creep; numerical simulation

CTOD fracture toughness test for super thick welded joints of D36 offshore platform steel

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Abstract: Based on BS7448 fracture toughness test experiment standard (ISO/TC164/SC4-N400) and DNV-OS-C401, crack tip opening displacement (CTOD) tests are carried out on 80 mm thick plates butt joints of offshore petroleum platform joined by the submerged-arc welding and CO₂ gas shielded welding. The CTOD values of weld metal and the heat-affected zone are tested at 0 °C. The possibility of avoiding post-weld heat treatment in thick welded joints is discussed using the submerged-arc welding and CO₂ gas shielded welding. These work provide the scientific basis on the offshore platform construction.

Key words: welded joints; crack tip opening displacement; fracture toughness

Mechanism and influencing factors of spatter formation with self-shielded flux cored wire

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Abstract: The spatter formation mechanism of self-shielded flux cored wire has a remarkable different with other welding materials because of its speciality in wire's construction and welding method. The mechanism and influencing factors of welding spatter are studied. High-speed photographs indicate that there are mainly three types of spatter in self-shielded flux cored arc welding, which are big spatter caused by arc force, bubble released spatter and gas explosion spatter. Five flux ingredients, which influence the welding spatter greatly, are investigated by means of spatter tests using mixing material regression method. Experimental result shows that the effect of fluorides and carbonates is greater than that of oxides, killing agents and alloys.

Key words: self-shielded flux cored wire; spatter; formation mechanism