# 不同组织超细化预处理下的异种钢超塑焊接

张柯柯¹,杨蕴林¹,赵宁²,王长生¹,王要利¹

(1河南科技大学 材料科学与工程学院,河南 洛阳 471003, 2西安交通大学 材料科学与工程学院,西安 710049)

摘 要:通过对 40C r/T10A 钢试样及试样待焊面表层分别实施盐浴加热循环淬火、高频淬火及激光淬火的组织超细化预处理,探讨了钢待焊表面组织对 40C r/T10A 异材恒温超塑焊(ISSW)工艺及接头质量的影响。试验结果表明,钢待焊表面组织对 ISSW 接头的形成有重要影响。待焊面组织越细, ISSW 所需焊接温度向低温区移动,初始应变速率向高应变速率区域移动, ISSW 所需时间越短;即使待焊双方一方实施组织超细化,也可实现接头抗拉强度达到 40C r母材的强度,但 ISSW 所需压接时间稍长。 ISSW 属小变形焊接,接头变形主要集中在原界面附近的淬火区,且 T10A 侧的变形均大于 40C r侧。

关键词: 钢; 组织超细化; 超塑焊接; 工艺; 接头质量 中图分类号: TG 453. 9 文献标识码: A 文章编号: 0253-360X(2006)01-37-04



张柯柯

## 0 序 言

固态焊接头区不经熔化与凝固过程,更易实现 与母材组织性能一致的高质量连接,这将成为 21世 纪有重要发展的连接技术<sup>[1]</sup>。恒温超塑性焊接(B SW)作为一种材料固态连接新技术,其工艺设备简 单, 易于实现精密焊接, 并可与塑性成形同步, 这是 现有压力焊接技术所不能比拟的,已成为近年来固 态焊接非常活跃的研究领域。尤其是超塑成形 扩散 连接(SPF IDB)在航空、航天、军工等行业的应用已 带来了巨大的经济效益,成为航空航天制造业中无 可替代的关键技术[2]。尽管人们对一些机械基础 件材料的 ISSW 工艺及接头形成过程进行了研究, 如结构钢 工具钢刀具、钢 铜合金发动机柱塞泵、 钢 钛合金仪表元器件及铜合金 钨合金电触头等的 ISSW<sup>[3~7]</sup>。但有关待焊材料组织在固态焊接过程 中的作用却鲜见文献报道,这严重地制约着对固态 焊本质的认识及固态焊新技术的开发。结构钢(如 40Cr) 工具钢(如 T10A)采用一般熔焊方法难以获 得良好的焊接接头。为此,作者以 40Cr/T10A ISSW 为例,探讨待焊面组织对其 ISW 的影响,以期为高 质量 ISSW 新方法的开发提供理论依据。

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# 1 试验材料及方法

试验用 40Cr和 T10A 钢为国产热轧退火态棒材,化学成分见表 1, 40Cr显微组织为珠光体 +铁素体,T10A 为细粒状珠光体,制成  $$^{\dagger}15$  mm  $\times$  25 mm 的试样。

表 1 试验用钢的主要化学成分(质量分数, %)
Table 1 Chemical compositions of steels

| 钢号    | С         | Мn               | Si               | Сг      |
|-------|-----------|------------------|------------------|---------|
| 40C r | 0 37~0 45 | 0 5~08           | 0. 17 ~ 0 37     | 0 8~1 1 |
| T10A  | 0 95~1 04 | $\leqslant 0$ 40 | $\leqslant 0$ 35 |         |

采用快速加热相变淬火方法在试验用钢试样表面获得 ISW 所需的超细化非平衡组织薄层。其中,整体盐浴加热循环淬火(SCQ), $40\mathrm{Cr}$  经  $820\,^{\circ}\mathrm{C}$  循环淬火 2 次, $T10\mathrm{A}$  经  $780\,^{\circ}\mathrm{C}$  循环淬火 3 次,淬火后在  $200\,^{\circ}\mathrm{C}$  回火  $30\,^{\circ}\mathrm{C}$  那 in,高频表面淬火  $(\mathrm{HFH}\,)$  是在  $\mathrm{GP100}\,^{\circ}\mathrm{C}$  到 高频感应加热装置上进行,淬火介质为聚乙烯醇水溶液,淬火后经  $2\,000\,^{\circ}\mathrm{C}$  回火  $30\,^{\circ}\mathrm{C}$  5 m in,激光表面淬火  $(\mathrm{LH}\,)$  采用  $5\,\mathrm{LW}\,$  的  $\mathrm{CW}\,^{\circ}\mathrm{C}$  00  $^{\circ}\mathrm{C}$  回火  $30\,^{\circ}\mathrm{C}$  3 mm, 搭接量为  $1\,\mathrm{mm}$ ,  $40\mathrm{Cr}\,^{\circ}\mathrm{C}$  1 描速度为  $3\times10^{-2}\,\mathrm{m}\,^{\circ}\mathrm{s}$  功率为  $1\,200\,\mathrm{W}$ ,  $T\,10\mathrm{A}\,^{\circ}\mathrm{H}$  描速度为  $1\times10^{-2}\,\mathrm{m}\,^{\circ}\mathrm{s}$  功率为  $9\,00\,\mathrm{W}$ ,淬火后经  $2\,00\,^{\circ}\mathrm{C}$  回火  $30\,^{\circ}\mathrm{C}$  3  $5\,\mathrm{m}$  in 经检测,超细化预处理后试样表面 和心部均获得回火马氏体,原奥氏体晶界用过饱和苦味酸钠溶液浸蚀显示,用截线法  $1\,\mathrm{m}$  1  $1\,\mathrm{m}$  3  $1\,\mathrm{m}$  4  $1\,\mathrm{m}$  3  $1\,\mathrm{m}$  4  $1\,\mathrm{m}$  3  $1\,\mathrm{m}$  4  $1\,\mathrm{m}$  5  $1\,\mathrm{m}$  4  $1\,\mathrm{m}$  5  $1\,\mathrm{m}$  6  $1\,\mathrm{m}$  6  $1\,\mathrm{m}$  7  $1\,\mathrm{m}$  8  $1\,\mathrm{m}$  9  $1\,\mathrm{m}$  8  $1\,\mathrm{m}$  9  $1\,\mathrm{m$ 

测得其晶粒平均截线长如表 2所示。显然,经 HFH 预处理后有更好的细化效果。将经超细化预处理或未处理的待焊面磨光 (表面粗糙度  $Ra<1~\mu m$ )并清洗干净,配对  $(40Cr/\Gamma10A$  异材 )对接置于压头速度可在 0.05~3.5~mm m in范围内连续可调的压接装置内,并施加  $\sigma_0=56.6~mm$  Pa预压应力。用控温精度为  $\pm 2~^{\circ}$ 0的 3~kW 电炉加热至焊接温度保温 10~mm in 在适宜初始应变速率下经短时压接后卸载空冷。

表 2 奥氏体晶粒平均截线长

Table 2 Average transversal length of austenite grain

| 预处理工艺  | 钢号    | 奥氏体晶粒平均<br>截线长 <i>1                                   </i> |  |
|--------|-------|--|--|
| 整体盐浴加热 | 40C r | 9 87   |  |
| 循环淬火   | T10A  | 8 63   |  |
| 高频表面淬火 | 40C r | 7. 18  |  |
| 同侧衣曲冲入 | T 10A | 6 72   |  |

将焊后的试样加工成  $\phi_{5\,mm} \times 25\,mm$  拉伸试样, 在 WE -300材料试验机上进行接头抗拉强度试验。用金相显微镜及 H -800透射电镜对焊接区进行显微组织观察和分析。

# 2 试验结果及分析

## 2.1 焊接变形特点

40Cr/T10A ISW 焊后试样宏观形貌如图 1所示。由图 1可见,ISW 压接后在靠近原界面处接头均发生鼓形变形,T10A 侧具有较大的径向变形 (膨胀率);与经 SCQ后压接试样变形有所不同,待焊面表层经 HFH 和 LH 后的压接试样变形主要集中在接头区附近,其余部位变形甚微;单侧经 HFH 和 LH后焊接试样的变形仅表现在近界面处预处理侧略有凸起。与 40Cr/T10A 真空扩散焊接约 1%的焊接变形相比,ISSW 变形较大,但明显小于变形焊。因此,ISSW 应属于低温小变形焊接。

#### 22 焊接工艺参数

大量试验结果表明, 压接双方或单方经超细化预处理 (包括 SCQ、H FH 和 LH)的 40C r/T10A ISSW 焊后拉伸试样, 拉伸试验中大部分断在原界面处。若焊接工艺参数合适, 则断口一般都断在距界面较远 (4~5mm 以外)的 40C r母材一侧, 反映出接头抗拉强度已达到甚至高于 40C r母材的强度, 其宏观断口呈典型的韧性断裂的杯锥状, 微观形貌以韧窝特征为主。

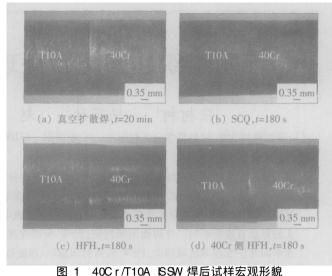


Fig. 1 Macro appearance of specimen after ISSW

基于组织超塑性的 ISSW, 焊前钢的组织由于影响材料的超塑性而影响焊接效果。具有不同组织超细化效果的 SCQ、HFH、LH三种预处理工艺对 ISSW 主要工艺参数的影响如下。

焊接温度,一般将压接双方的超塑性温度重叠区间作为其焊接温度,适宜的焊接温度既要满足金属超塑性流变的要求又要兼顾原子的扩散。焊接温度与接头强度的关系见图 2 由图 2可见,预处理后组织细化效果愈好,则达到 40℃ 和母材强度(经HFH和 IH 预处理 ISW 焊后 40℃ 基材抗拉强度为560~570 M Pa)的 ISW 所需的温度向低温区移动。经 SQQ 预处理的焊接接头抗拉强度要明显高于经HFH和 IH 预处理的抗拉强度约 100 MPa 主要是由于前者整个试样具有淬火 +高温回火态的

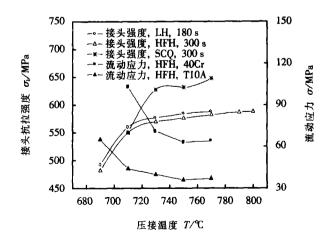


图 2 焊接温度与接头强度的关系

Fig. 2 Relationship between we tling temperature and tensile strength of joint

(预压应力  $\sigma_0 = 56.6 \text{ MPa}$  初始应变速率  $\epsilon_0 = 2.5 \times 10^{-4} \text{ s}^{-1}$ ; 应变  $\epsilon = 0.15$ )

组织,而后者淬火区以外为供应态的退火组织。在拉伸试验时,同样断在接头区外的母材 40Cr处,前者显然具有较大的抗拉强度。在相同超塑压缩变形条件下 T10A比 40Cr具有较小的超塑流动应力(超塑均匀变形时对应的流动应力,见图 2 应变  $\varepsilon=0$  15)和较大的应变速率敏感性指数 m,见式(1)、 $(2)^{(9)}$ ,这是导致不同预处理情况下压接试样的 T10A 侧径向变形(膨胀率)高于 40Cr侧径向变形(膨胀率)的原因。

初始应变速率,一般处在待焊双方最佳超塑应变速率范围的上限。初始应变速率与接头强度的关系见图 3 由图 3 可见,焊接前组织的细化效果愈好,则达到 40Cr母材强度 ISW 的实际初始应变速率向高应变速率区域移动。

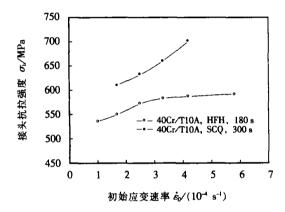


图 3 初始应变速率与接头强度的关系 Fig 3 Relationship between initial stain rate and tensile strength of joint (预压应力  $\sigma_0 = 56.6 \, \mathrm{MPa}$  压接温度  $T = 750 \, ^{\circ}\mathrm{C} \,)$ 

压接时间,在 ISSW 过程中,接头强度与时间的 变化规律均近似于幂规律(图 4)。如果压接时间过 短,接合面就会残存空隙致使焊接接头断在接口处。 从图 4和表 3可见,一般需几分钟即可使接头强度 达到 40C r母材的强度,延长压接时间对接头强度几 乎无影响,这与一般需要几十分钟的扩散焊接过程 相比要短得多。由 SCO→HFH→LH, 40Cr/T10A 异 材 ISSW 接头达到 40Cr钢母材强度所需时间明显 缩短 (由约 360 s→约 180 s→约 90 s), 这佐证了晶界 在超塑性及 ISSW 中所起的重要作用<sup>[2]</sup>,即随着焊 接前组织的细化,以晶界滑移为主的材料超塑性效 应更明显,其流动应力 σ减小、应变速率敏感性指 数m 值升高、超塑压缩变形激活能 $Q_c$  降低,见式 (1)、(2);另一方面较细的晶粒为原子提供了更多 的快速扩散的 "通道 ",扩散系数越大[2]。因此,焊 前组织的超细化,有利于形成冶金结合所需的塑性 流变和扩散,因而更容易实现焊合。

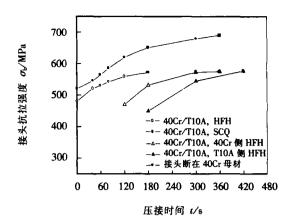


图 4 ISSW 过程中接头强度变化 Fig. 4 Tensile strength of joint in process of ISSW (预压应力  $\sigma_0=56$  6M  $P_{a}$  压接温度 T=750  $^{\circ}$ C, 初始应变速率  $\epsilon_0=2.5\times10^{-4}~{\rm s}^{-1}$ )

表 3 不同工艺方案达到 40C r母材强度所需时间(s)
Table 3 Times obtained to strength of 40C r base
metal in different technical projects

| SCQ - | HFH   |            |          | TH   |
|-------|-------|------------|----------|------|
|       | HFH   | 40Cr侧 H FH | T10A侧HFH | m    |
| 360   | 约 180 | 300        | 420      | 约 90 |

SCQ: 
$$\begin{cases} \sigma_{1} = e^{-18.57} \varepsilon^{0.34} \exp(\frac{209.9 \times 10^{3}}{RT}), \\ \sigma_{2} = e^{-17.29} \varepsilon^{0.41} \exp(\frac{193.8 \times 10^{3}}{RT}), \end{cases}$$
(1)  
HFH: 
$$\begin{cases} \sigma_{1} = e^{-17.44} \varepsilon^{0.36} \exp(\frac{199.6 \times 10^{3}}{RT}), \\ \sigma_{2} = e^{-16.15} \varepsilon^{0.48} \exp(\frac{183.6 \times 10^{3}}{RT}), \end{cases}$$
(2)

式中:  $\sigma_1$ 、 $\sigma_2$ 分别为 40C r钢、T1 0A 钢的流动应力; m 值、 $Q_c$  值均为所选温度区间 (730 ~770  $^{\circ}$ C)的平均值;  $\varepsilon$ 为  $\varepsilon_0$  /(10<sup>-4</sup> s<sup>-1</sup>).

与 ISW 相比, U - ISW (单边或单侧恒温超塑性固态焊接)可降低待焊双方之一的组织要求,尤其是对接头强度要求不太高时,它不仅简化了预处理工艺,更重要的是它可使一些不具备组织超塑性的材料实现 ISW,具有开发应用价值。从图 4和表 3可见,由于单边或单侧超塑性,欲达到相同焊接效果, U - ISW 比 ISW 需要更长时间;在 ISW 过程中,40C r一侧的超塑效应对焊接效果具有更大的作用。

但在实际应用中,为实现较精密的 ISSW,需要控制变形量  $\varepsilon$ ,而  $\varepsilon \approx \varepsilon$  t 因此为获得可靠的 ISSW 接头,可选用不同的压接时间 t和应变速率  $\varepsilon$  搭配,如较高的  $\varepsilon$  下压接时间短些、较低的  $\varepsilon$  下时间长些,

但在焊接生产中应用还是前者更适合。

### 23 接头区组织

通过对 40C r/T 10A 钢 ISW (含 U – ISW)接头显微组织形貌特征的观察和分析,认为接头中存在 '界面超细晶区'、'过渡区 '等组织特征区域。

界面超细晶区,是指在压接前的升温、保温和焊接过程中,界面接触处的两侧金属受塑性变形和超塑性变形作用而发生动态再结晶现象形成的等轴、细小的公共晶粒构成的冶金结合区域(见图 5)。 界面接触处微区所受应力状态的不同,导致其塑性变形状态的不同而使焊接面各微区焊合状态是非均匀的,该区域一般 $\leq 2 \mu_m$ 。



图 5 接头区界面处的 TEM 照片 Fig 5 Interface microstructure of joint zone

过渡区,在界面两侧附近与远离界面的母材之 间有一明显的组织过渡区。主要指在超塑效应作用 下, 近界面处 (两侧 )形成的扩散区, 该区域的组织 形态包括其中的碳化物数量、形态和分布等都发生 了明显变化。 T10A 侧, 由于 C 向 40C r侧扩散而使 C含量降低,在近界面 2~3个晶粒宽的范围内,出 现了平均尺寸为  $2 \sim 3 \mu_{\rm m}$  且晶内几乎无碳化物、近 于等轴状的  $\alpha$  晶粒 (图 5右上方区域), 其显微硬度 明显较低 (该区域硬度约 200 HV 0 1); 40C r侧, 近 界面处组织均为 α晶粒及珠光体 (α基体内片状与 粒状碳化物共存)的组织形态, α晶粒平均尺寸明显 较小(约  $0.5\mu_{\rm m}$ ), 珠光体含量明显高于远离界面 的 40C r母材, 尤其是近界面的原 α //相界及三叉晶 界珠光体量更多。这表明 T10A 侧 C 原子向 40Cr 扩散, C的扩散使  $\alpha$ 中 C浓度升高即导致  $\gamma$ 含量的 增高而使珠光体量增多,这充分反映了晶界是原子 扩散的快速通道。焊接前组织越细,ISSW 与 U -ISSW 相比, 具有更宽的过渡区。

## 3 结 论

- (1) 钢待焊表面组织对 ISSW 接头的形成有重要影响。焊接前钢的组织越细, ISSW 所需焊接温度向低温区移动, 初始应变速率向高应变速率区域移动, ISSW 所需时间越短; 即使待焊双方一方实施组织超细化, 也可实现接头抗拉强度达到 40Cr母材的强度, 但 ISSW 所需压接时间稍长。
- (2) ISSW 属小变形焊接。接头变形主要集中在原界面附近的淬火区,且 T10A 侧的变形均大于40Cr侧。
- (3) ISSW 接头中存在着明显的 '界面超细晶区 '、'过渡区 '等组织特征区域。

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作者简介: 张柯柯 男, 1965年2月出生, 工学博士, 教授, 国际焊接工程师(WE)。主要从事新材料连接与特种连接技术、微电子连接用无铅钎料及其焊接性等方面的研究工作。先后主持参与完成省部级科研项目10余项, 发表论文70余篇。

Email zhkeke@mail haust edu. cn

ture toughness test standard.  $\delta$  – Resistance curve tests were conducted with multiple specimens at 5  $^{\circ}$ C in welded joints of X56 pipeline steel. And then the  $\delta_{0.2}$  values of weld and HAZ were obtained. The assessment was carried out by using the maximum stress at different water depths and the stress concentration and residual stress were also considered. The toterable surface flaw sizes were gained at perfect a lignment and maximum allowable m is a lignment. This study lays the foundation for judging the acceptability of flaws

**Keywords** British standard tolerable size engineering critical assessment

Finite element analyses of instantaneous stresses of hot plate welled joint of plastic pressure pipes WANG Jian ping<sup>1</sup>, HUO Li xing<sup>1</sup>, GU Kan feng<sup>2</sup> (1. School of Materials Science and Engineering Tianjin University Tianjin 300072 China 2 Shenyang Institute of Automation Chinese Academy of Sciences Shenyang 110015 China). p21 – 25

Abstract B ased on constitutive relationship of them alviscoelastic integral mode considering the properties of material which depend great on temperature fluctuation and the effect of phase transition latent heat heat force coupling function and load step character of ANSYS were employed to simulate the hot plate welding procedure of high density polyethy lene(HDPE) plastic pressure pipe. The finite element analyses on stresses distribution of welded joint was conducted and transient stresses distribution pattern in axial radial and circum ferential directions were obtained. Residual stresses was measured by the math armethod and the sachs method and it showed that the actual measured results of the residual stresses are basically coincident with the results of numerical analysis.

Keywords high density polyethy lene, pressure pipe, hot plate, welded join; instantaneous stress, finite element analyses

#### Effect of single component fluoride flux on TIG arc shape for Tial

by LIX iao hong ZHANG Lian feng DU Yu xiao Beijing Aeron au ticalM anu facturing Technology Research Institute Beijing 100024 China). p26 – 28

Abstract BT20 titanium alloy of 2.5 mm thickness was welded by conventional TIG and single component fluoride flux TIG respectively. The results showed that the influences of different single component fluorides on the shape of arc are different. And the relationship of arc shape and the weld penetration was also obtained.

 $\begin{tabular}{ll} \textbf{Keywords} & single component fluoride flux} & A-TIC welding & arc shape \\ \end{tabular}$ 

Edge detection of weld in age based on Canny operator XIE Zhimeng GAO Xiang dong(Department of Mechanical and Electrical Engineering Guangdong University of Technology Guangzhou 510090 China). p29 - 32 36

Abstract Weldedge is the most important characteristic in aweld

image. The fact of selecting an appropriate edge detecting operator and obtaining the accurate well edge information is a key process in well image processing. The canny operator was used to extract the well edge and its principle and implementation were discussed. And the canny operator kirsch operator prewitt operator robert operator sobel operator and gauss laplace operatorwere applied in detecting the well edges to test the effects respectively. The analyzsis of the well image and well center coordinates showed that the canny operator is a very effective way to detect the well edge and also suitable for the seam tracking process based on vision sensors.

**Keywords** Canny operator edge detection, seam tracking image processing

Numerical sinu lation of improving welling efficiency of TIG surface protection ZHENG Shu<sup>1</sup>, HU Guo hui<sup>1</sup>, CHEN Fang quan<sup>2</sup>, YANG Huanming<sup>3</sup> (1. Shanghai Institute of Applied Mathematics and Mechanics Shanghai University Shanghai 200072. China 2. College of Manufacturing Automation. Shanghai University. Shanghai 200436. Chrina 3. Jian Zhong Chemicals Corporation. Sichuan Yibin 644009. Chrina). p33 – 36

Abstract Based on the Navier Stokes equation and renomalization group (RNG) & & turbulence model the flows around the nuclear fuel stick and the weld cavity were simulated numerically by the SM-PLEC algorithm. The body fitted coordinates staggered grid method and self-adaptive technique was utilized in this study. Through the analysis of flow around the nuclear fuel stick and the weld cavity—the defect of the old structure was found out—and two methods to improve the welding efficiency were proposed.

Key words computational fluid dynamics nuclear fuel stick SMPLEC algorithm

Superplastic solid state welding of dissinilar steels under different microstructure by ultra fining treatment. ZHANG Kekel, YANG Yurn lin<sup>1</sup>, ZHAO Ning<sup>2</sup>, WANG Chang sheng<sup>1</sup>, WANG Yao li<sup>1</sup> (1. Material Science and Engineering College Henan University of Science and Technology Henan Luoyang 471003 China 2 School of Material Science and Engineering Xián Jiaotong University Xián 710049 China).

Abstract Them icrostructures of 40C rand T10A steel and its surfaces were ultra fined through salt bath cyclic quenching high frequency hardening and laser hardening then influence of the welded surface microstructure of steels on the process and joint quality of isothermal super plastic solid state welding (ISSW) for 40C r/T10A dissimilar steels was studied. The experimental results showed that the welded surface microstructure of steel plays an important role in the formation of BSW joint. The welding temperature of BSW extends to bw temperature ranges and the initial strain rate extends to higher strain rate ranges while having finer

we kled surface microstructure of steel before welling. If only one surface is eligible for ultra-fining, the tensile strength of the joint will be also up to that of 40°C r base metal, but the welling time is slightly longer. ISSW belongs to the little deformation welling the deformation of joint ismainly located in the quenched area near original interface. The deformation of T10A side is larger than that of 40°C r side.

Keywords steel microstructure ultra fining treatment super plastic solid state welding process

Phase transform a tion diffusion bond ing technology for titan ium alby to stain less steel QN Bin<sup>1</sup>, SHENG Guang min<sup>1</sup><sup>2</sup>, HUANG Jirwei<sup>1</sup>, LI Cong<sup>2</sup> (1 College of Material Science and Engineering Chongqing University Chongqing 400044 China 2 National Key Laboratory for Nuclear Fuel and Material Nuclear Power Institute of China Chengdu 610041 China). p41 – 44, 48

Abstract The joints of titanium alloy (TA17) and stainless steel (OCr18N  $\theta$ Ti) were obtained by phase transformation diffusion bonding Effect of parameters on strength of the joint was investigated and the optimum parameters for bonding are as follows maximum cyclic temperature is 890 °C, number of cyclic times is 10 bonding pressure is 5 MPa and heating velocity is 30 °C/s Strength of the joint under the optimum condition is 307 MPa and the time for bonding is 160 s. Scanning electron microscopy (SEM), energy dispersive spectroscopy (EDS) and X-ray diffraction (XRD) were used to study the fracture appearance of the joint. The study showed that the fracture takes place at somewhere between FeTi and  $\beta$ -Ti layers, and the FeTi layer is the weak est point in the joints. The joint was analyzed by EDS and the ternary phase diagram for FeCrTi. The results indicated the presence of  $\sigma$ , Fe2 Ti FeTi and  $\beta$ -Ti in the reaction zone between stain less steel and titanium alloy.

Keywords titan immalloy stain less steel phase transformation, diffusion bonding

M etal transfer of twin wire indirect arc argon welding CAOMei qing ZOUZeng da DUBao shuai QUShiyao WANGXir hong LI De gang(School of Material Science and Engineering Shandong Univer sity Jinan 25006l, China). p45 – 48

Abstract The metal transfer and the connesponding are voltage and welding current in twin wire indirect are argon welding was investigated with high speed camera system based on a xenon lamp source and digital oscillogragh. Results showed that with the different matching of welding current and are voltage the mode of metal transfer consists of short circuiting transfer globular transfer mixing transfer projected transfer and streaming transfer etc. With the increasing of welding current the droplet size reduces and the droplet is refined. With the raising of are voltage the size of droplet reduces. There is well corresponding relation ship between metal transfer mode and the oscillogram of volgtage and

welding

Keywords indirect are; metal transfer short circuiting transfer streaming transfer

Autonom ous seam tracking based on local vision in arc robotic welding ZHOU Lii, CHEN Shan ben, LN Tao CHEN Wen jie (Welding Institute Shanghai Jiaotong University Shanghai 200030 China). p49 – 52

Abstract A method of autonomous seam tracking in arc robotic welding was presented. It could make the robot get rid of traditional teaching and playing back mode and automatically acquired the special coordinates of welding seam. Them aster computer continuously processed local area images in front of the torch in the images which were acquired by a CCD camera to measure the deviation between the torch and the seam and the orientation of the seam. At the same time, the computer controlled the robot to make the torch move forward along the seam and recorded the seam seam of coordinates in the robotic basic reference frame which were corrected by the deviations. After the torch arrives at the end-point of the seam, the computer controlled the robot to move the torch to the first record coordinate and begin to well. Experiments were made on the curves seam workpieces of mild steel and aluminum alloy work pieces.

Key words in age processing are welling robost autonomous seam tracking local vision

Distribution of Au during reaction of eutectic SnPb solder and Au /

Ni/Cu pad LIF tr quan<sup>1</sup>, WANG Chun qing<sup>1</sup>, DU Miao<sup>2</sup> (1 National Key Laboratory of Adran ced Welding Production Technology Harbin Institute of Technology Harbin 150001 China; 2 Harbin Welding Institute Harbin 150080 China). p53 – 56

Solder bump was fabricated with Sm-Pb eutectic solder Abstract droplet on Au Ni/Cu pad. The solder /pad was then subject to reflow soldering and aging at 125 °C. The MC evolution at solder/pad interface during this process especially the formation and distribution of Au-Sn compound were investigated. The results showed that Au-Sn compound forms at solder/pad interface during contact reaction and Au does not re act fully with solderdroplet During the subsequent reflow soldering all Au layer at interface is consumed. Ni layer reacts with solder which leads to the formation of Ni<sub>3</sub>Sn<sub>4</sub> compound at the interface. Acicular AuSn<sub>4</sub> can be found in the solder bulk. AuSn<sub>4</sub> particles redesposites at the interface as a continuously (  $\mathrm{Au_{r}N}\,i_{\mathrm{l}\,\mathrm{-x}}$  )  $\mathrm{Sn_{4}}\,$  layer during aging at 125 °C. The redesposited (Au<sub>x</sub>N i<sub>1-x</sub>) Sn<sub>4</sub> at solder /p ad interface follows decomposition diffusion mechanism. At the same time a lead rich phase emerges with AuSn<sub>4</sub> redeposition at the interface. The shear strength of soldered joint is mainly determined by this evolution and distribution of Au Sn compound.

Keywords solder bump Au Ni Cu, reflow, ageing in terme