

5A90 铝锂合金激光焊焊缝微观组织特征

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摘 要: 采用 Nd YAG 激光焊进行了 5A90 铝锂合金薄板的对接试验, 测试了接头横截面不同位置沿水平方向的显微硬度. 利用光学显微镜、扫描电镜 (SEM) 和透射电镜 (TEM) 分析了焊缝金属不同位置的显微组织变化. 结果表明, 焊缝区不同位置的显微硬度低于母材; 焊缝熔合线附近出现了等轴晶粒细晶区 (EQZ); 5A90 合金主要的析出强化相是 δ 相, 在焊缝金属不同位置都有 Li_2 超点阵衍射斑点的出现, 证实焊缝中有 δ 相析出, 但数量很少, 这解释了焊缝显微硬度降低的原因.

关键词: 铝锂合金; 激光焊; 显微组织; 析出相

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0 序 言

铝锂合金由于具有低密度, 高比强度, 良好的耐腐蚀性和卓越的超塑成形性能, 用其取代常规铝合金, 可使构件质量减轻 10%~15%, 刚度提高 15%~20%^[1], 被认为是 21 世纪航天航空工业最理想的轻质高强结构材料. 在航空航天构件的制造中, 用焊接工艺代替铆接等机械工艺可以减轻结构质量, 提高运载能力, 获得显著的经济效益^[2]. 传统的弧焊热输入量大, 变形严重, 接头强度较低, 制约了这类材料尤其是薄板结构应用的进一步扩大. 激光焊接具有热输入小, 变形小, 接头性能优良等优点, 有望解决现有工艺存在的问题, 已被证明是实现这类材料连接的先进方法^[3]. 但是, 激光焊接头区域的组织与母材相比有很大的改变, 导致铝锂合金激光焊结构整体性能和寿命的下降.

5A90 铝锂合金是一种 AlLiMg 系新型国产化合金, 是前苏联 1420 铝锂合金的相近牌号, 具有中等强度、良好的抗腐蚀性和出色的焊接性^[4]. 但是, 目前有关 5A90 铝锂合金激光焊接接头的微观组织特征未见公开报道. 文中以 5A90 合金薄板激光焊接头为对象, 采用金相显微镜、扫描电镜 (SEM) 和透射电镜 (TEM) 对焊缝的显微组织进行分析, 探讨激光焊对铝锂合金接头微观组织的影响, 可为合理

制定 5A90 铝锂合金激光焊工艺提供依据, 也将为铝锂合金激光焊在航空器结构中的应用奠定基础.

1 试验方法

试验材料为国产 5A90 铝锂合金, 厚度为 3 mm. 其化学成分如表 1 所示. 激光焊接系统采用 AMB56 型 Nd YAG 激光器, 最大输出功率 4.5 kW, 激光连续输出, 光束质量参数 (BPP) 为 25 mm·mrad. 传输激光的光纤直径为 0.6 mm, 聚焦透镜焦距为 200 mm. 焊前用化学方法去除试件表面的油脂和氧化膜. 焊接试验采用平板对接方式, 焊接速度为 2.7 m/min, 激光功率 1.8 kW, 激光离焦量 0 mm. 焊接过程中采用高纯度氩气对焊接区进行背保护和侧吹保护.

表 1 5A90 铝锂合金化学成分 (质量分数, %)

Table 1 Chemical composition of 5A90 AlLi alloy

Mg	Li	Zr	Fe	Si	Ti	Al
4.5~6.0	1.8~2.3	0.08~0.15	<0.2	<0.15	<0.05	余量

焊后线切割试板, 经过镶样、研磨、抛光和腐蚀, 制成金相试样. 使用 JEOL2010 透射电镜观察显微组织, 工作电压为 200 kV. 电镜试样使用 1:3 的硝酸甲醇溶液双喷穿孔制备 TEM 样品薄膜. 在显微硬度仪上测试接头的显微硬度, 加载 0.98 N, 持续时间 15 s.

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2 试验结果

2.1 焊接接头的显微硬度

图 1 是 5A90 铝锂合金激光焊接头横截面不同部位沿水平方向的显微硬度分布。从图 1 可知, 5A90 合金的显微硬度在 125 HV 左右, 而焊缝区的显微硬度值平均在 90 HV~110 HV 之间, 焊缝区的显微硬度低于母材的硬度。此外, 焊缝中部的显微硬度平均为 109 HV 顶部硬度平均值达 107 HV 底部平均硬度值只有 103 HV 三层的硬度差别不大, 但焊缝中部的显微硬度最高。

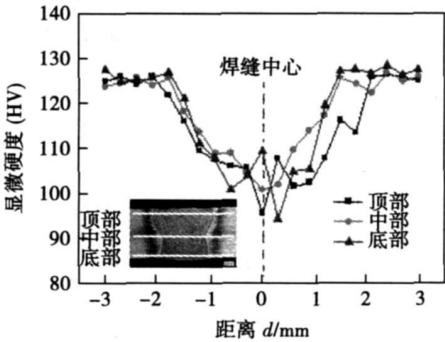


图 1 焊缝横截面的显微硬度

Fig 1 Microhardness distribution in transverse section of weld

2.2 光学显微组织观察

如图 2 所示, 5A90 铝锂合金激光焊接头由母材、热影响区 (HAZ)、细晶区 (EQZ) 及焊缝 (WM) 组成, EQZ 位于 HAZ 与 WM 之间, 阻碍外延柱状晶的生长, 各部分对应的高倍光学照片见图 3。母材具有明显轧制特征的流线组织。在图 2 中特别光亮的部分为 HAZ 如图 3 所示, 可观察到拉长的扁平状晶粒, 尺寸在 30~300 μm 之间。在熔合线附近存在一个由非常精细的等轴晶粒组成细晶区 (EQZ), 晶粒

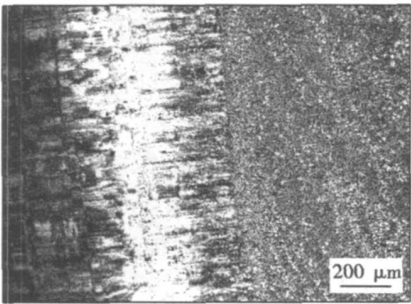
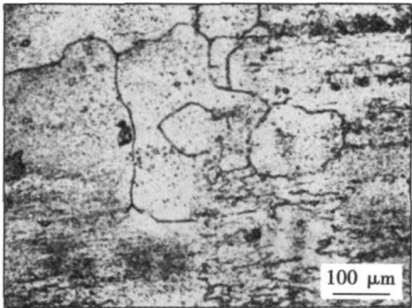


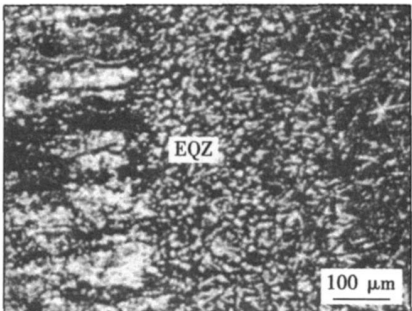
图 2 接头整体形貌

Fig 2 Whole configuration of joint

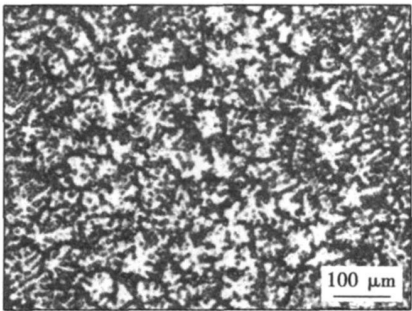
尺寸约 5~10 μm 之间, 如图 3 b 所示。EQZ 已经在 1420 8090 2090 等铝锂合金钨极氩弧焊接头中发现。EQZ 是铝锂合金焊缝组织中的特有现象, 与一般情况下焊缝熔合区组织不同, EQZ 与母材之间没有明显的联生结晶的特征。EQZ 的形成与焊缝液态金属中合金元素 L 和 Z 元素对熔池结晶的作用有关^[5 6]。



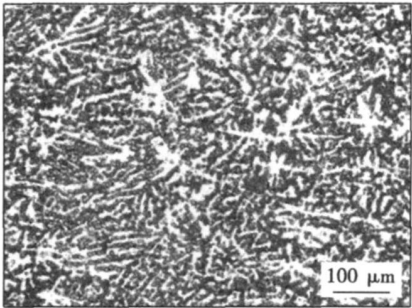
(a) 接头热影响区的显微组织



(b) 接头熔合线附近的 EQZ 组织



(c) 焊缝中的等轴晶



(d) 焊缝中枝晶发达的等轴晶

图 3 激光焊接头的显微组织

Fig 3 Microstructure of welded joint

5A90合金以杂质的悬浮质点形核,焊缝组织呈现出独特的组织特征,即从熔合线至焊缝中心,结晶形态依次为细晶区、柱状晶区和等轴树枝晶区.在EQZ形成之后,EQZ附近温度梯度 G 最大、而结晶速度 R 最小,不易满足成分过冷的形成条件.新相晶核只能以细晶带晶粒作为现成表面形核,晶粒取向处于最有利生长方向上,即与最大散热方向一致时的晶粒优先成长,向焊缝中心生长形成柱状晶组织,但由于细晶带的存在,柱状晶的尺寸也得到了细化.在一般铝合金焊缝金属中,在 G/R 大的焊缝边缘长出柱状晶,其亚结构依次为平面晶、胞状晶和树枝晶,在 G/R 小的焊缝中部,则由于独立形核产生等轴树枝晶^[7].但在 5A90 铝锂合金激光焊缝中,改变了上述凝固组织形态变化规律.焊缝中的柱状晶区较小,较大区域的柱状晶被等轴晶取代,并且在焊缝中由于熔池波动还会有层状线的生成(图 2),由二次枝晶发达的等轴树枝晶和细小等轴晶两种形态组成,如图 3c、d所示.焊缝等轴晶区面积相对较大则是由多种因素造成的.铝锂合金含有较多的 Li 元素,致使熔池金属的临界形核功显著降低,大大促进了异质形核;Li 作为表面活性元素易于在界面前沿的液体中富集并吸附在晶核表面,一方面阻碍了微小晶体的成长,使晶粒细化,另一方面则导致成分过冷增大,促进新相成核.此外,元素 Zr 能与 Al 元素形成高熔点化合物 Al_3Zr 在液态铝中, Al_3Zr 能够作为铝液中异质形核所需的现成界面,促进异质形核^[8].

2.3 SEM观察

采用 SEM对焊缝顶部、中部和底部中心线位置的组织形态进行了观察,如图 4所示,焊缝组织主要是以 α (Al) 固溶体为主,在基体晶界和枝晶界分布着大量的 $\alpha + \delta$ (AlLi) 共晶组织,已被腐蚀呈凹坑.由图 4还发现,焊缝不同位置的组织形态不同:在顶部及中部的晶界、枝晶界呈短棒状分布,而在底部共晶相呈长条网络状分布,并且晶界、枝晶界的间距较大.已有证明,如果晶界上的共晶相呈网络状连续分布,焊缝金属的强度将降低^[9].焊缝中部的晶界、枝晶呈短棒状分布,并且界间距最小,因而其对应的显微硬度较高.

在扫描电镜下,观察母材及焊缝不同位置的试样,放大 1.5 万倍观察了母材及焊缝组织晶内的形貌.图 5a是母材 5A90 合金组织晶内的形貌,晶内弥散分布着大量球状相,尺寸大小在 20~200 nm 之间.这些粒子可能是 δ (Al_3Li) 相.图 5b~d分别为焊缝顶部、中部和底部组织晶内的形貌.与母材相比,焊缝组织晶内的形态与母材差别很大,焊缝组织

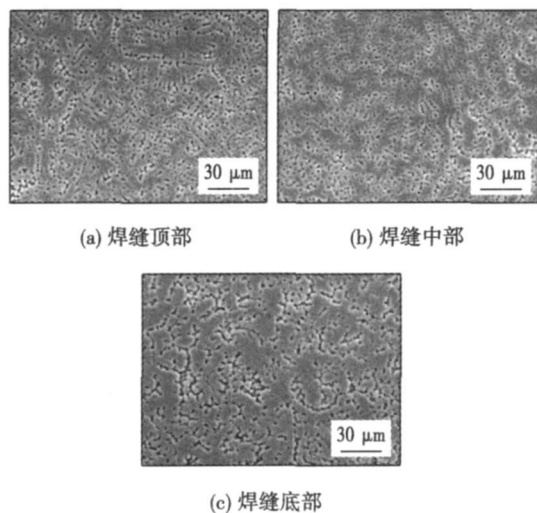


图 4 焊缝区的 SEM 组织形貌
Fig. 4 SEM morphology of weld zone

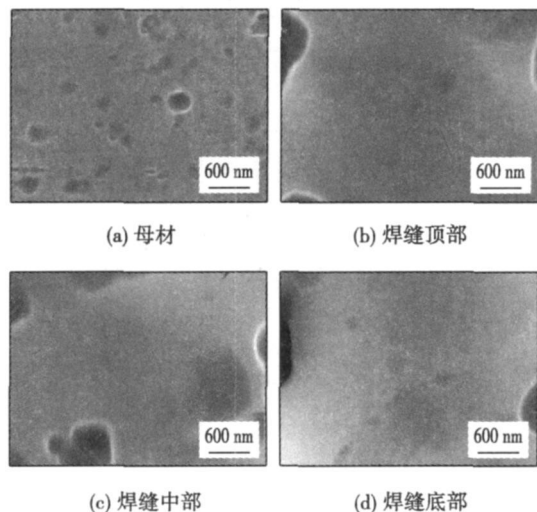


图 5 焊缝与母材的析出相形貌
Fig. 5 Precipitate morphology of weld and base metal

晶粒粗大,晶内的球状相数量明显减少.

2.4 TEM观察

图 6a、b为母材 5A90合金基体 $[112]$ 带轴选区电子衍射花样及其对应显微组织的明场像.基体 $[112]$ 带轴选区电子衍射花样主要呈现 L_2 超点阵衍射斑点(图 6a),其对应的明场像显示出晶内弥散分布着较多的球状粒子,尺寸大小约在 20~200 nm 之间,如图 6b所示.因此,这种球状粒子为 δ 析出相.这证实了 SEM观察到的球状粒子为 δ 析出相.此外,在 TEM下 5A90合金不仅观察到了上述的 δ 相,还观察到了另一种细小的析出相,其明场像如图 6c所示,这种粒子尺寸仅在 5~20 nm 之间,在晶内弥散均匀分布,基体 $[112]$ 带轴选区电子衍射花样与图 6a相同,也呈超点阵衍射斑点,证实这

种尺寸非常小的粒子也是 δ 相,但未能在 SEM 中观察到.在合金塑性变形时, δ 相尺寸相对较小,被运动位错切割,能够有效阻碍位错移动,从而增加合金的强度;当 δ 相尺寸较大时,不能被位错切割而只能被绕过,强化作用下降.此外,有文献报道, Al_2LMg 相也是 AlMgLi 系合金的另一种重要的相,在淬火和时效初期即可形成,呈针状或棒状,它的数量与分布和合金的耐腐蚀性能有关^[19].但试验中,未发现这种 Al_2LMg 相.因此,TEM 选区电子衍射花样分析表明 5A90 合金显微组织有尺寸大小不同的 δ 析出相.

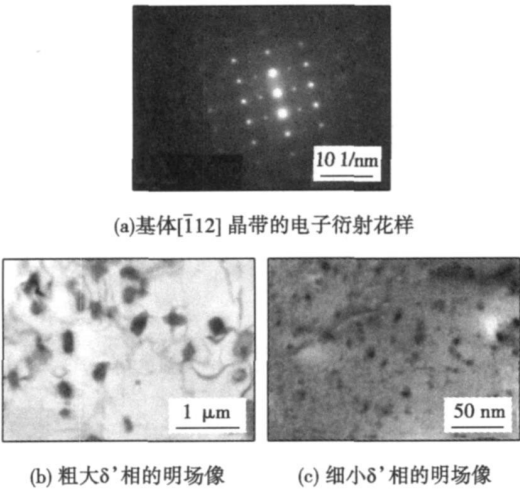


图 6 5A90 铝锂合金中的析出相
Fig 6 Precipitates in 5A90 Al-Li alloy

图 7 为 5A90 合金激光焊焊缝不同位置试样的选区电子衍射花样,发现 δ 相超点阵斑点,证明焊缝中存在 δ 相.但其对应的暗场像观察均未发

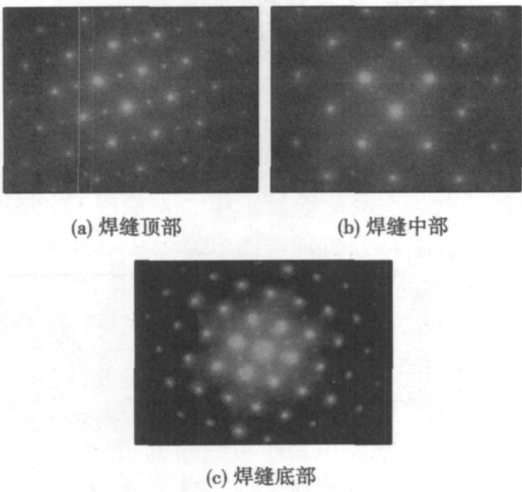


图 7 焊缝 TEM 选区电子衍射花样

Fig 7 Diffraction Patterns in selected area from weld

现有 δ 相的析出,这说明焊缝中有 δ 相,但数量极少,这与焊缝 SEM 的观察结果一致.焊缝中的 δ 析出相较少导致了焊缝时效的不足.激光焊接时冷却速度很快,由于时间很短而析出相数量很少,析出相强化不足,从而造成了焊缝区的软化^[11].

3 结 论

- (1) 5A90 铝锂合金激光焊焊缝熔合线附近存在等轴晶粒细晶区 (EQZ),焊缝区由两种不同形态的等轴树枝晶组织组成.
- (2) 5A90 铝锂合金激光焊焊缝硬度低于母材,存在接头软化的现象,焊缝顶部、中部及底部的平均硬度值差别不大.
- (3) 5A90 合金中有尺寸大小不同的 δ 析出相,而焊缝中只观察到 δ 相的超点阵衍射斑点, δ 析出相数量很少,造成了焊缝时效的不足,这导致了焊缝硬度的降低.

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(σ_{cr}) 以下时将不会发生波浪变形, 即消除了薄板的失稳的根本因素。

根据焊接失稳变形及旋压后变形量的测量结果(表 4)可知, 焊件 1、2、3 和 4 表示随焊旋转挤压在不同的工艺参数影响下挠度变化的情况。

表 4 焊接失稳变形及旋转挤压后变形量
Table 4 Original buckling and deformation degree after revolution extrusion

焊件	焊后挠曲高度 h_1/mm	加致力 F/N	杆枪距 l/mm	加载后挠曲高度 h_2/mm
1	17	1 600	25	13
2	17	1 600	25	14
3	17	1 900	25	10
4	17	1 900	25	8

常规焊接件在不加随焊处理时, 其挠曲高度为 17 mm 左右。焊件 1 和 2 表示距焊枪 25 mm 处进行旋转挤压, 挤压的压力为 1 600 N。随焊旋转挤压后其挠度由常规焊接件的 17 mm 下降到 13 mm 和 14 mm。从焊件 3 看出, 当焊枪距不变的情况下, 加大旋转挤压力, 其焊件的挠度将继续下降。说明当随焊旋转挤压的压力增加时, 薄板变形的挠度呈现出下降的趋势, 但是挠曲度仍然较大。这是由于加载发生塑性拉伸变形后, 焊缝在随后的冷却过程中仍然会由于冷却收缩而受到拉应力的作用, 导致与之平衡的残余压应力上升, 当达到临界失稳应力时, 发生失稳变形。当然在高温阶段的加载使得残余压应力的峰值降低, 这将在部分程度上抵消挠曲度。

3 结 论

(1) 通过随焊旋压的焊接工艺, 在一定程度上减少 TC4 焊接过程发生失稳变形的程度, 由原来的 17 mm 下降到 13 mm 或 8 mm。即通过在焊缝的高

温区施加一定量的载荷使该区域发生塑性延展来抵消冷却过程的压缩变形, 使得原有的挠曲度下降了 1/3 左右。

(2) 随焊旋压工艺中随着载荷水平的增加, 其变形的程度将随之下降, 即在此工艺中施加的压力将会对焊接的失稳变形起到关键性的影响。

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Abstract The characteristics of heat source for vacuum electron beam welding of deep penetration were analyzed. On the basis of the Gaussian descending for heat source radius and exponential increasing for power density of heat source along the depth, a mathematical model of whirling Gaussian body source with increasing peak power was established. Under the level of focusing and defocusing, the power density distribution for special thermal effect of vacuum electron beam welding was calculated by this model. The results of these calculations showed that the power density distribution of the heat source model was consistent with the thermal characteristic of electron beam source based on the idealized state. And this model was applicable to the thermal effect simulation of vacuum electron beam welding at different focus levels.

Key words electron beam welding; body source; power density

Microstructure investigation of Nd:YAG laser welded 5A90 aluminum-lithium alloys CUI Li, LIX Jiaofan, HE Dinyong, CHEN Li, GONG Shuif (1. College of Materials Science and Engineering, Beijing University of Technology, Beijing 100124, China; 2. High Energy Density Beam Processing Technology Department, Beijing Aeronautical Manufacturing Technology Research Institute, Beijing 100024, China), P 77—80, 84

Abstract Nd:YAG laser welding of 5A90 aluminum-lithium (Al-Li) alloy thin sheets has been carried out. Microhardness tests across the transverse section of the weld were performed. The influence of laser welding on the microstructure of 5A90 aluminum-lithium alloy at different sections has been evaluated by optical microscope, SEM and TEM. Microhardness tests revealed that the laser weld had lower microhardness than that of base metal. Laser welds revealed the formation and presence of a zone of equiaxed grains (EQZ) along the fusion boundary. The major strengthening phase in 5A90 aluminum-lithium alloy was δ phase particles. The presence of the δ precipitates in the weld was proved by the Li_2 superlattice reflections in diffraction patterns, but the volume of the δ precipitates in the weld zone was very little, which attributed to the decrease in the microhardness of the weld compared with the base metal.

Key words aluminum-lithium alloys; laser welding; microstructure; δ precipitates

Welding TC4 thin plates by revolution pressing in welding process ZHANG Yong², YANG Janguo, LIU Xuesong, FANG Hongyuan (1. State Key Laboratory of Advanced Welding Technology Production, Harbin Institute of Technology, Harbin 150001, China; 2. School of Material Science and Engineering, Liaoning Technical University, Fuxin 123000, Liaoning, China), P 81—84

Abstract The welded residual compression stress which is larger than the critical compression stress will cause buckling distortion of TC4 thin plate after normal welding. In order to decrease buckling distortion, the method of revolution pressing in welding process was utilized to gain plastic extension in the high temperature zone of the welding bead. The residual stress can be reduced by compensating the plastic shrinkage in the high temperature zone. Tensile test, SEM analysis of the tensile testing specimen and measurement of the joints distortion after welding were carried out. The results showed that the deflection could be

decreased to two thirds of the original deflection after the high temperature zone metal received revolution pressing. With increased pressing loading, the deflection reduced. In a word, the method of revolution pressing in welding process is feasible to control the deflection of TC4 thin plate weldment.

Key words revolution pressing in welding process; TC4; buckling distortion; thin plate welding

Electron beam brazing repair of K465 Ni-base superalloy blades WANG Gang, CHEN Guoqing, ZHANG Binggang, FENG Jicai, LU Chenglai (1. State Key Laboratory of Advanced Welding Production Technology, Harbin Institute of Technology, Harbin 150001, China; 2. AVIC Shenyang Limited Aero Engine (Group) Limited Corporation, Shenyang 110043, China), P 85—88

Abstract The vacuum electron beam brazing (VEBB) of K465 Ni-base superalloy blades with self-made filler metal was studied. Effects of through slot (grinding off all the cracks) and non-through slot (grinding off 80% to 90% the cracks) on the cracks of joints were investigated. It's discovered very few through slot specimens have cracks and all non-through specimens have cracks. Microstructure of the brazed joints with self-made filler metal was studied by means of scanning electron microscopy (SEM), energy disperse spectrum (EDS) and X-ray diffraction (XRD). The results show that the structure of brazing seam consists of Ni-base γ solid solution, Ni_3Si , Ni_3B , Ni_3Al and Ni_3Si .

Key words electron beam brazing; Ni-base superalloy blades; interfacial reaction product

Effect of laser power on welding process stability of Nd:YAG laser short circuit MAG hybrid welding WANG Wei, LIN Shangyang, WANG Xuyou, LEI Zhen (1. Harbin Welding Institute, Harbin 150080, China), P 89—92

Abstract The effect of laser power on welding process stability and the formation of weld were investigated during Nd:YAG laser short circuit MAG hybrid welding in this paper. The results showed that laser power was an important welding parameter which could affect welding process stability. When short circuit MAG welding process combined with laser power, welding process would become stable. But it was not true that welding stability would rise with laser power increased. The most stable welding process presented when laser power was lower. So according to the laser power, Nd:YAG laser short circuit MAG hybrid welding could be classified into three stages: the stabilizing arc stage with lower laser power, transition stage and deepening weld penetration stage with higher laser power. Finally, the reasons for how laser made arc to be more stable were analyzed.

Key words laser welding; arc welding; hybrid welding; GMAW

Effect of Al content on properties of Zn-Al filler metal ZHANG Ma², XUE Songhai, DAI Wei, LOU Yinbiao, WANG Shufeng (1. College of Materials Science and Technology, Nanjing University of Aeronautics and Astronautics, Nanjing 210016, China; 2. College of Mechanical Engineering, Huaiyin Institute of Technology, Huai'an 223001, Jiangsu, China; 3. Zhejiang Xinru Welding Material Limited Corporation, Shengzhou 312000, Zhejiang, China), P 93—96